

*Investigation of Damage Evolution, and  
Modeling of Residual Stress and Fracture  
Toughness in the Ti-Al<sub>3</sub>Ti Metal  
Intermetallic Laminate (MIL) Composites*

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**E.A. Olevsky (SDSU)**

## *Acknowledgements*

- Professor M.A. Meyers (UCSD)
- Professor E.A. Olevsky (SDSU)
- Committee members
  - Professor David J. Benson
  - Professor Lea M. Rudee
  - Professor Satchi Venkataraman
- JDP program between SDSU and UCSD
- DARPA

# *Outline*

- Introduction
- Research Objective
- Experimental Results
- Theoretical Modeling
- Preliminary Conclusions

# *Introduction*

## *Intermetallics - Applications*

- ❖ Aerospace structure
- ❖ Protective shields
- ❖ Automobile industry
- ❖ Coatings
- ❖ Armor

## *Intermetallics*

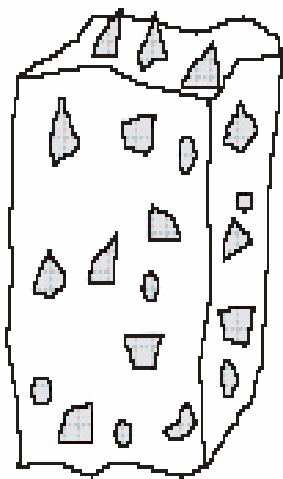
- **Advantages:**

- ▣ Good high-temperature strength
- ▣ High stiffness
- ▣ Good creep resistance
- ▣ High oxidation resistance

- **Disadvantages:**

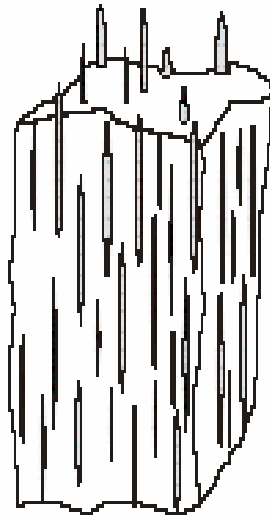
- ▣ Low tensile ductility at low temperature
- ▣ Poor fracture resistance

# *Toughening Intermetallics – Ductile Reinforcement*



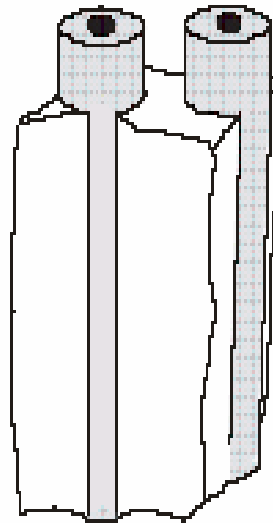
(a)

**Particle**



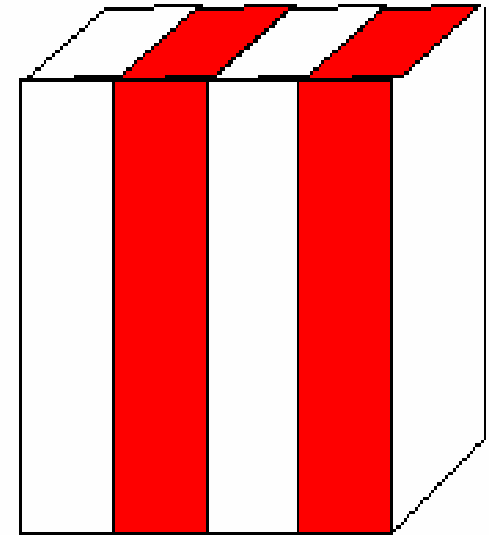
(b)

**Short fiber/Whisker**



(c)

**Continuous fiber**



(d)

**Layer**

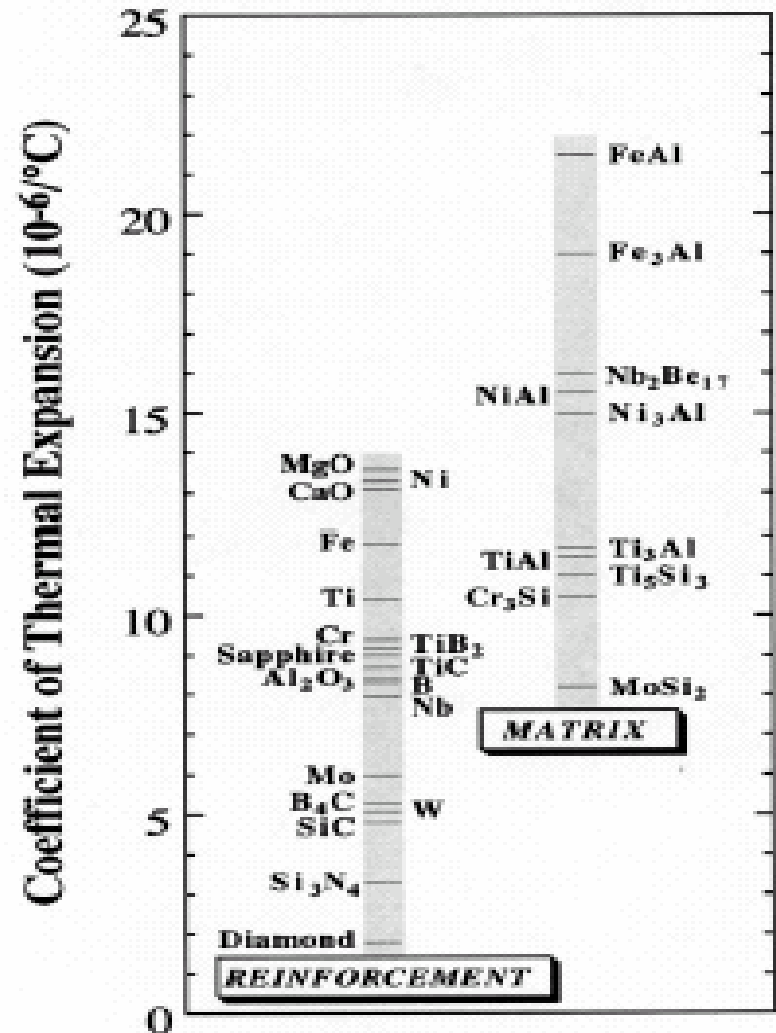
# *Challenges in Producing Ductile Reinforced Intermetallic Composites I*

- Chemical compatibility
  - To avoid undesirable microstructure and properties
- Environmental resistance
  - Subject to dynamic environment at low/intermediate temperature
- Consolidation and processing
  - Complete densification



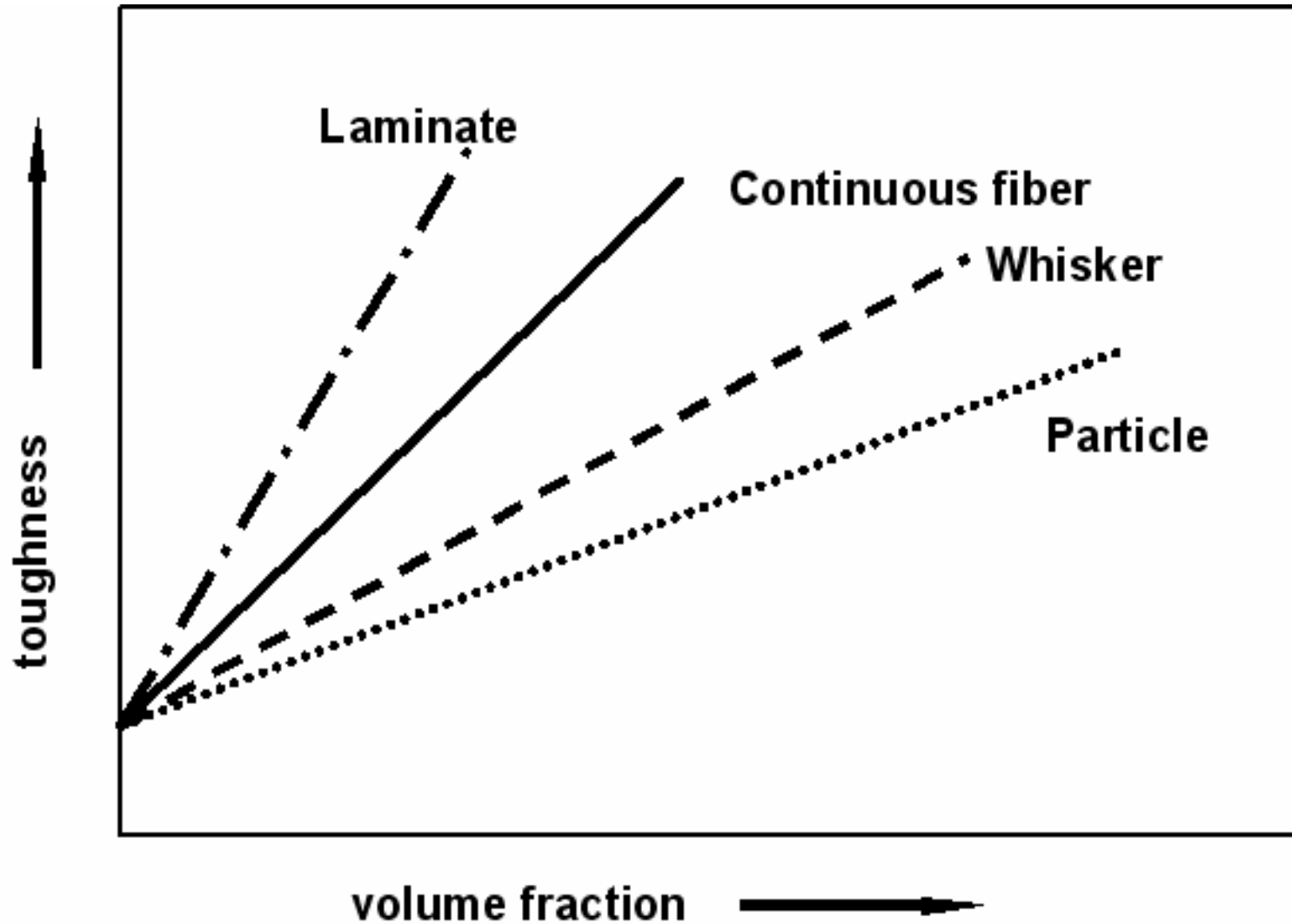
# Challenges in Producing Ductile Reinforced Intermetallic Composites II

- Mismatch of thermal expansion coefficients
  - Introduce residual stress during processing and service

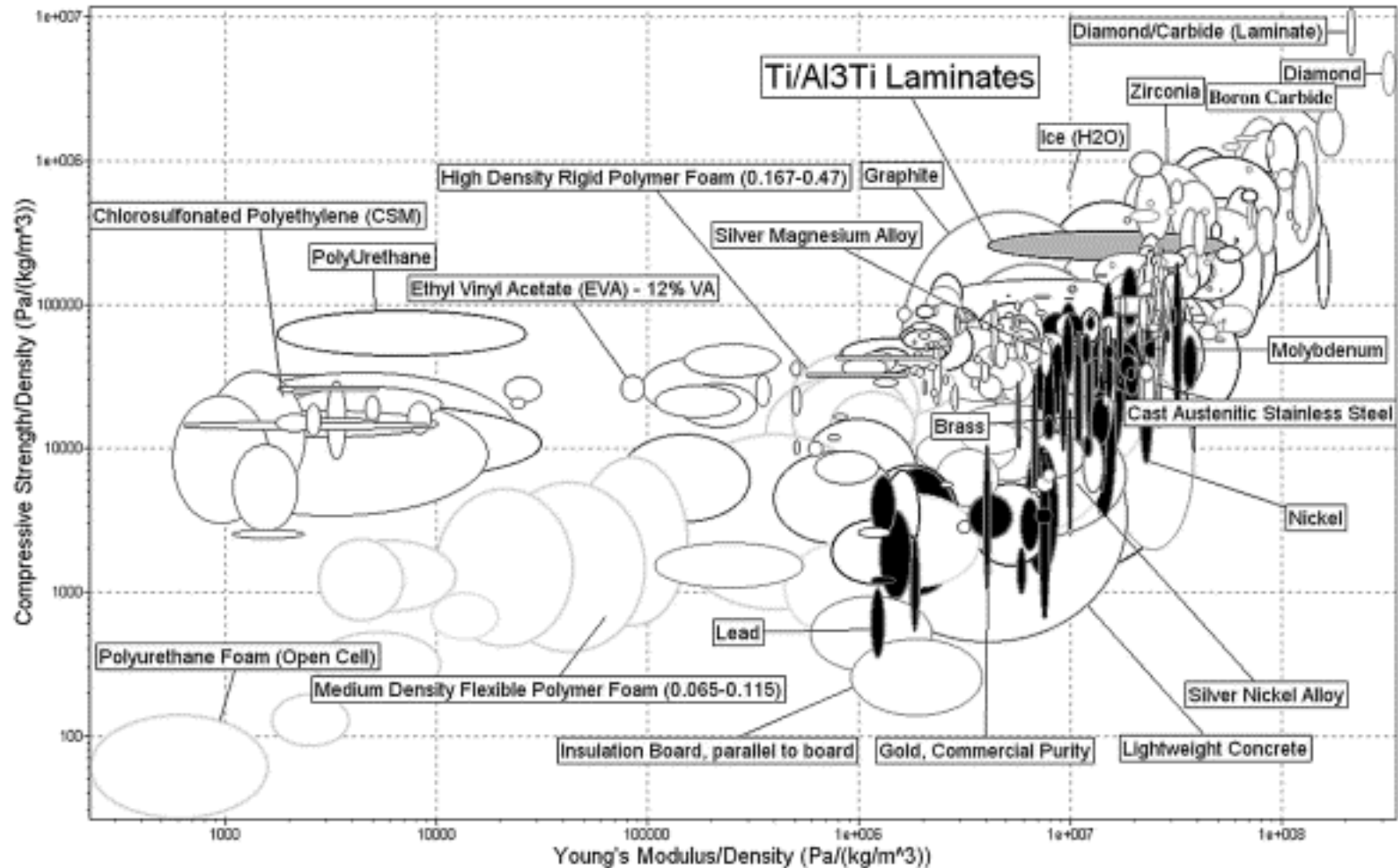


Courtesy to Miracle

## *Ductile Reinforced Laminate Composites*

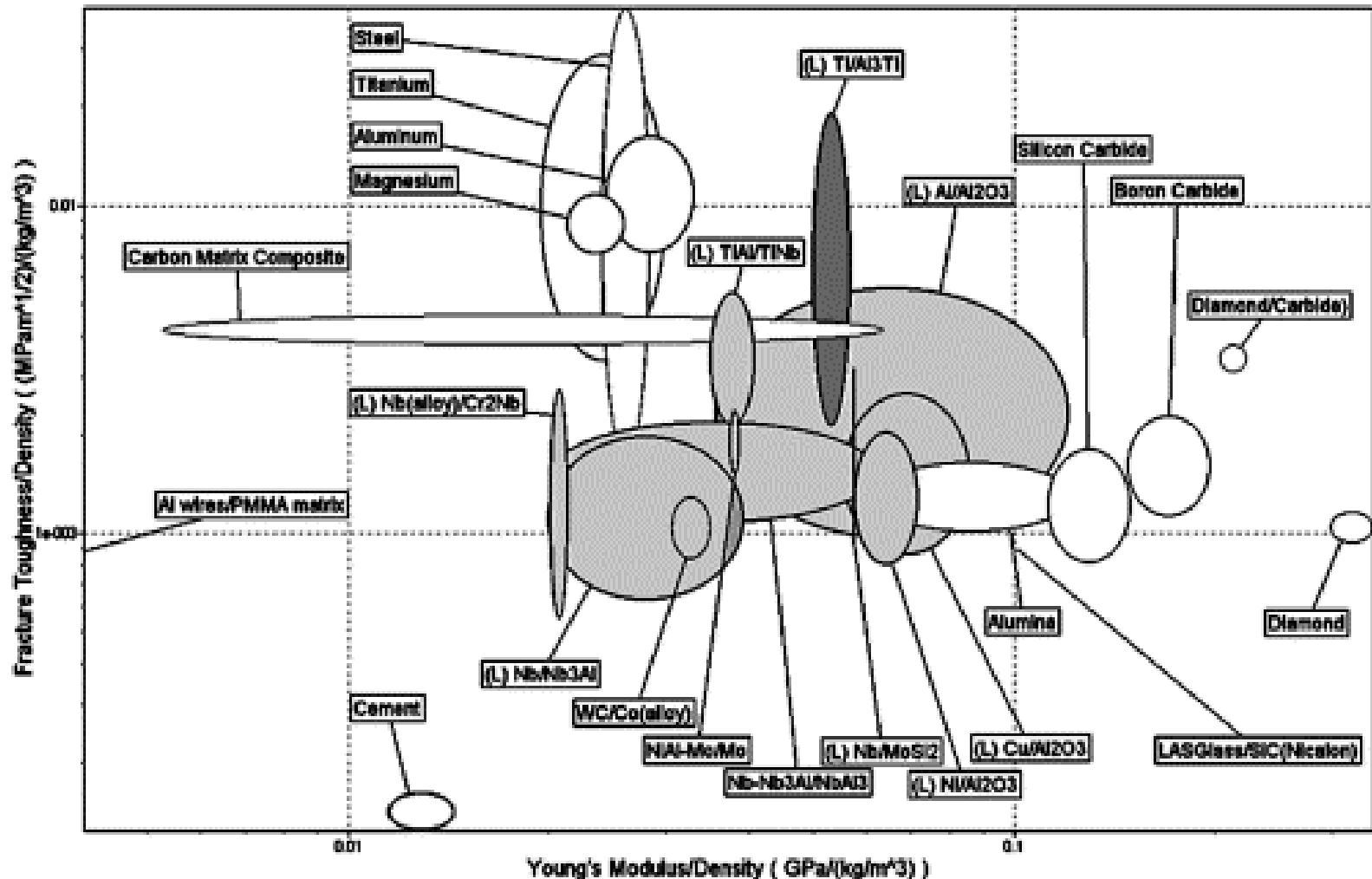


# *Ti-Al<sub>3</sub>Ti MIL Composites I*



Courtesy to Ashishy

## *Ti-Al<sub>3</sub>Ti MIL Composites II*



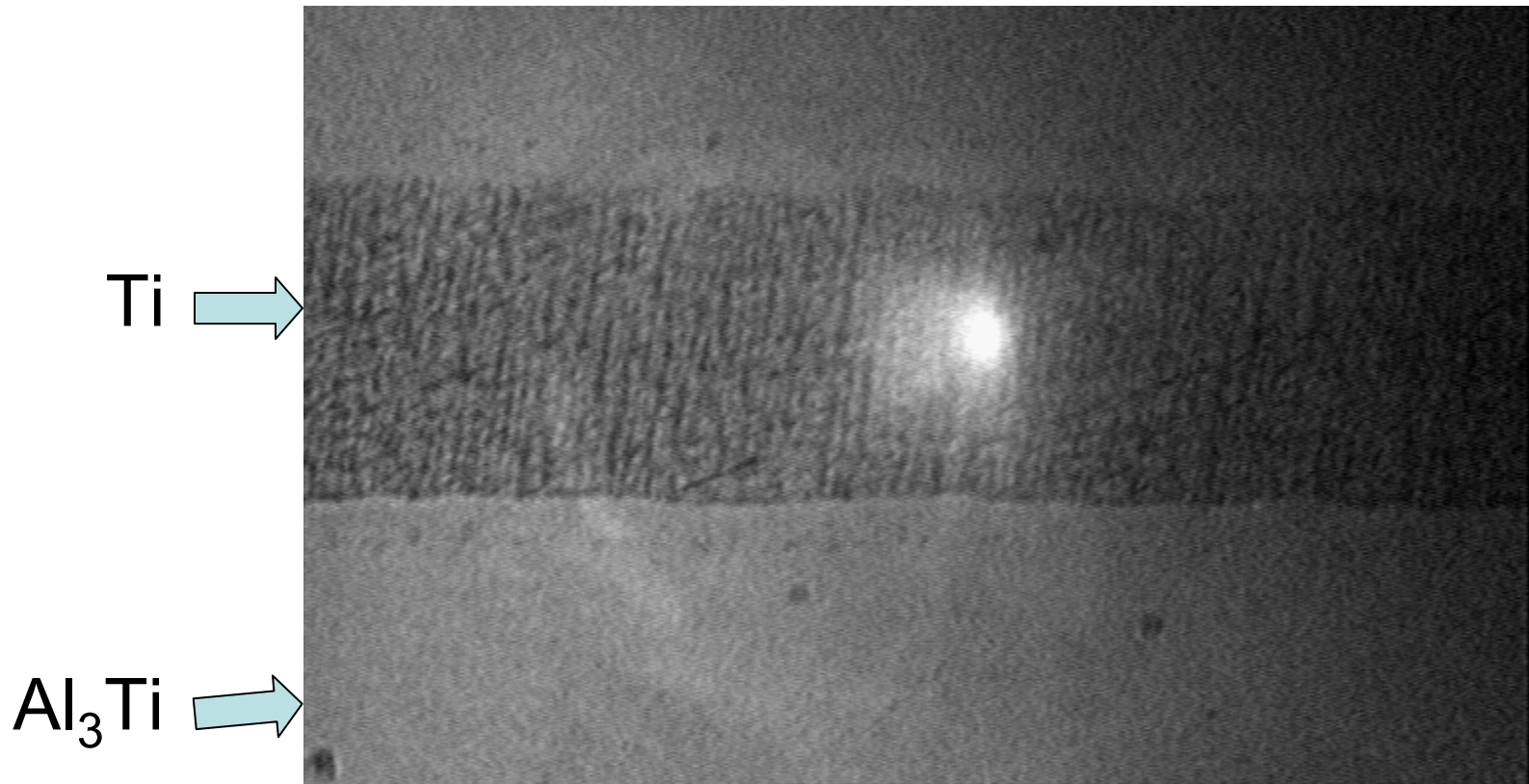
Courtesy to Ashishy

## *Research Objective*

- Investigation of damage evolution in Ti-Al<sub>3</sub>Ti metal intermetallic laminate (MIL) composites
- Modeling of residual stress and fracture toughness in Ti-Al<sub>3</sub>Ti metal intermetallic laminate (MIL) composites

# *Experiments*

# *Optical Microscopy Observation on Untested Ti-Al<sub>3</sub>Ti*

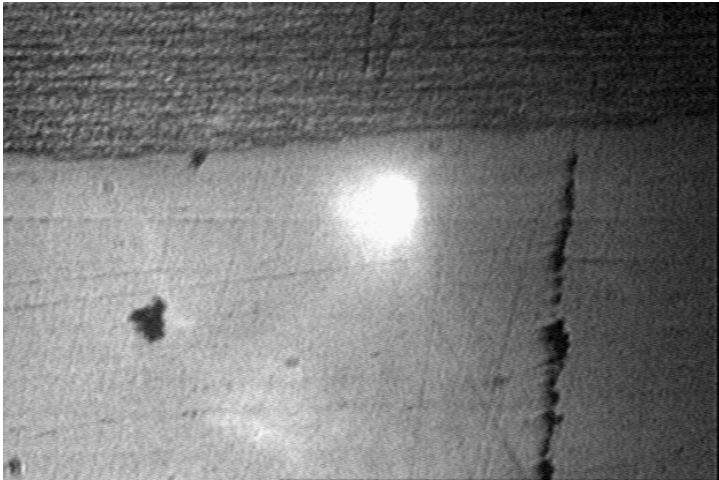


**Typical untested sample**

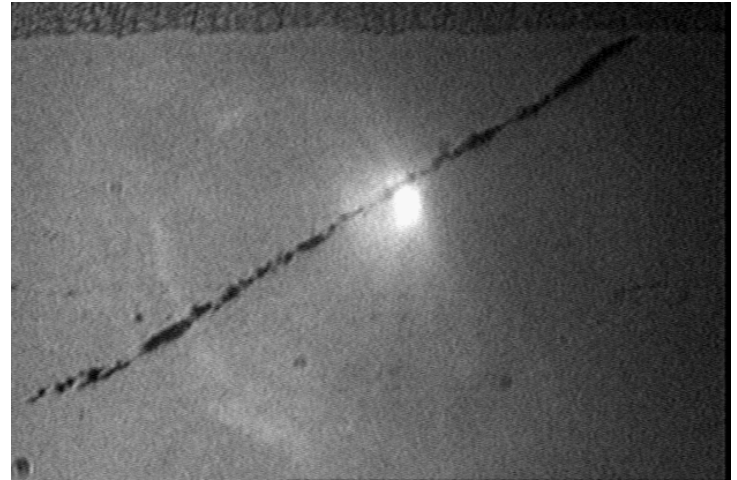
## *Cracks in Untested Ti-Al<sub>3</sub>Ti*



**Parallel cracks**



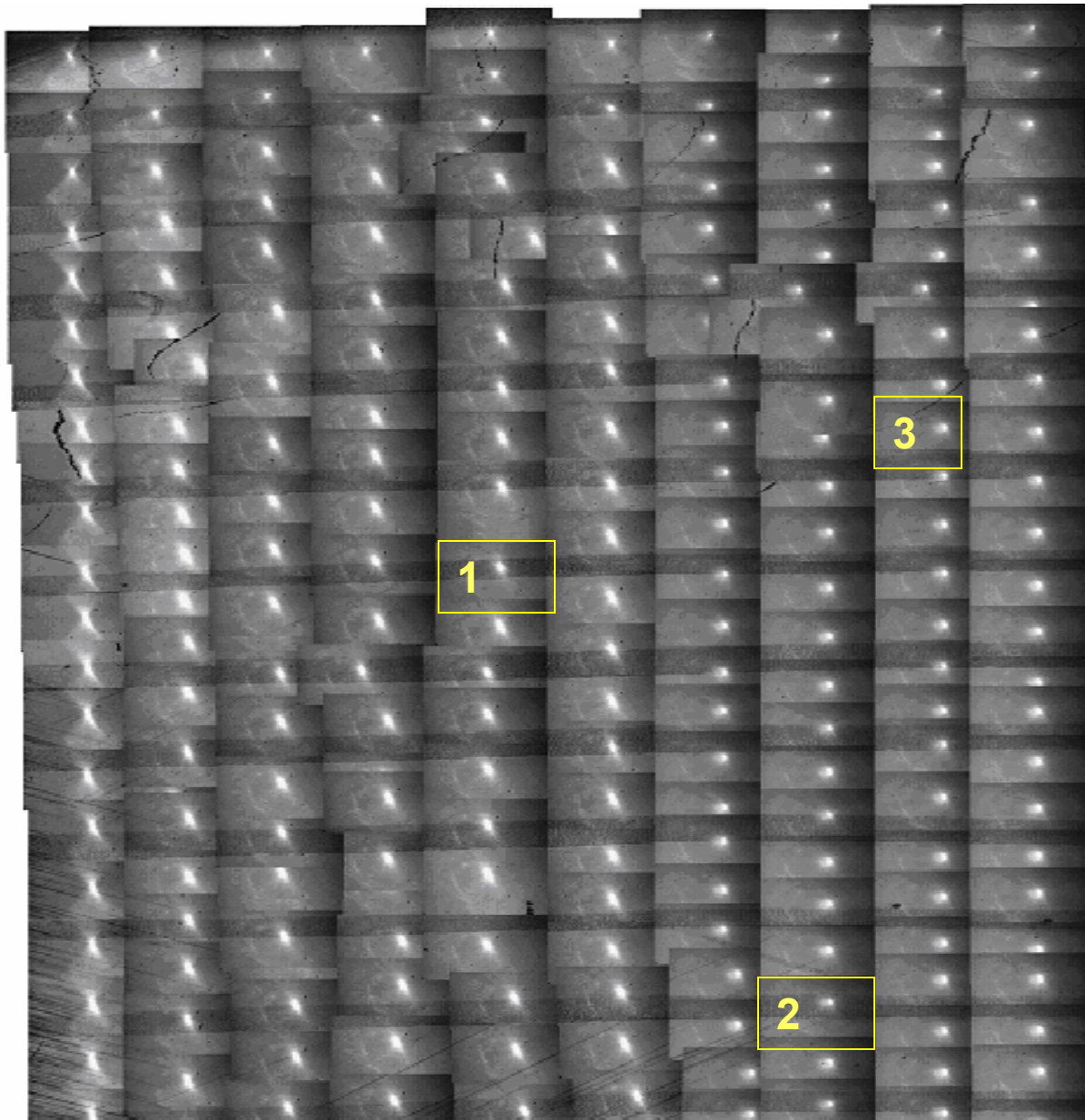
**Perpendicular cracks**



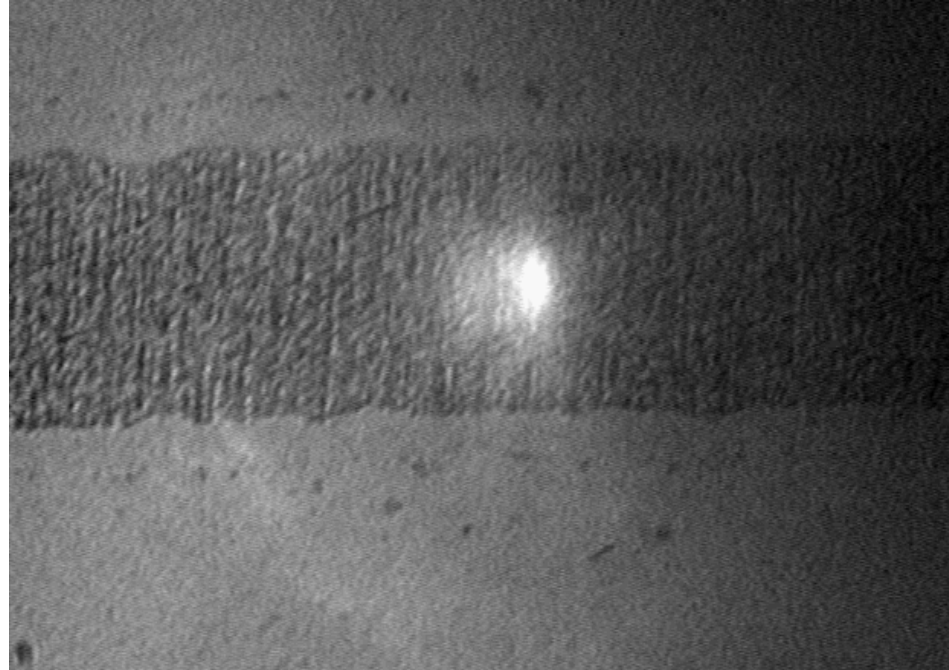
**45° angled cracks**



# *Optical Microscopy Observation on Untested Ti-Al<sub>3</sub>Ti*

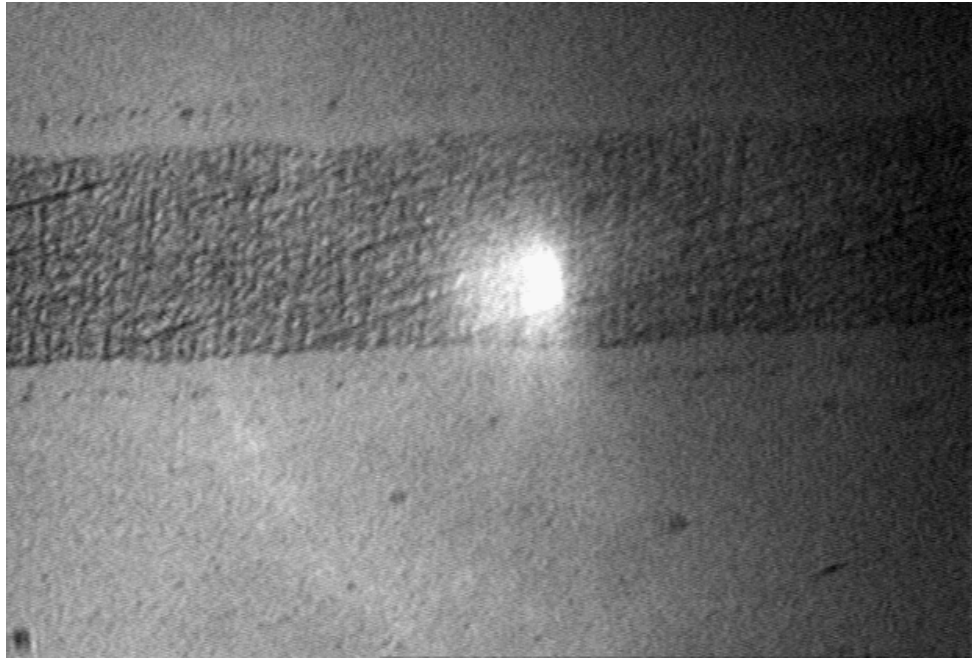


## *Micro-defects in Untested Ti-Al<sub>3</sub>Ti*



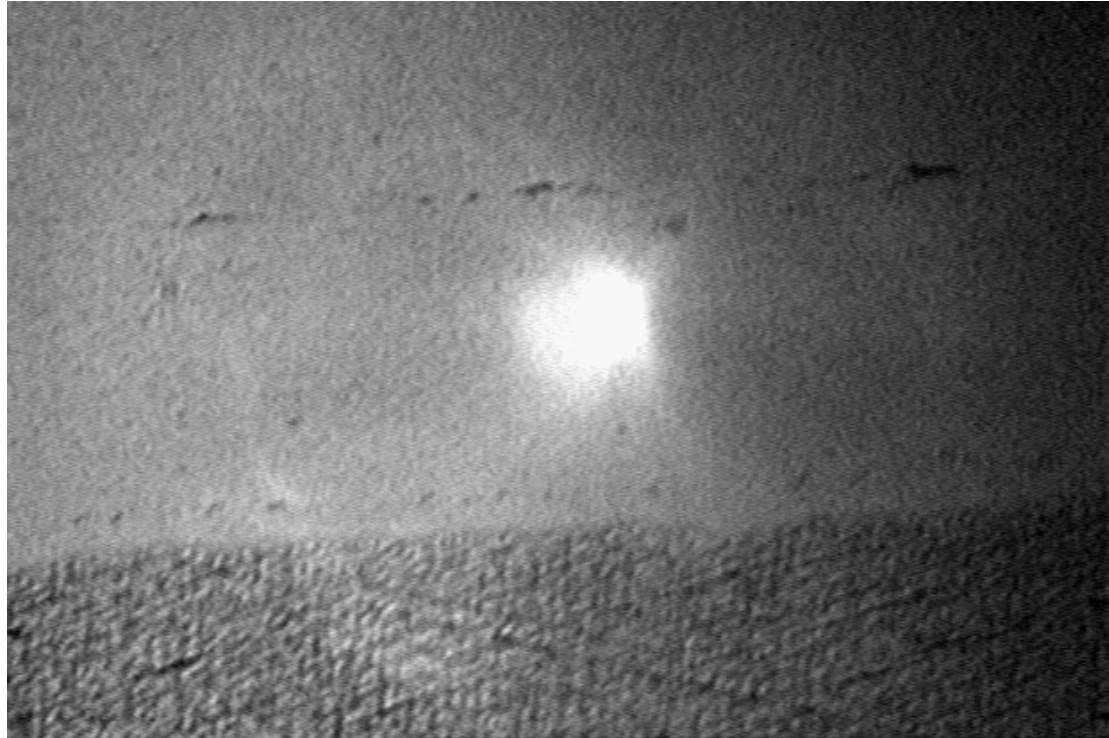
**Micro-defects along the interface**

## *Micro-defects in Untested Ti-Al<sub>3</sub>Ti*



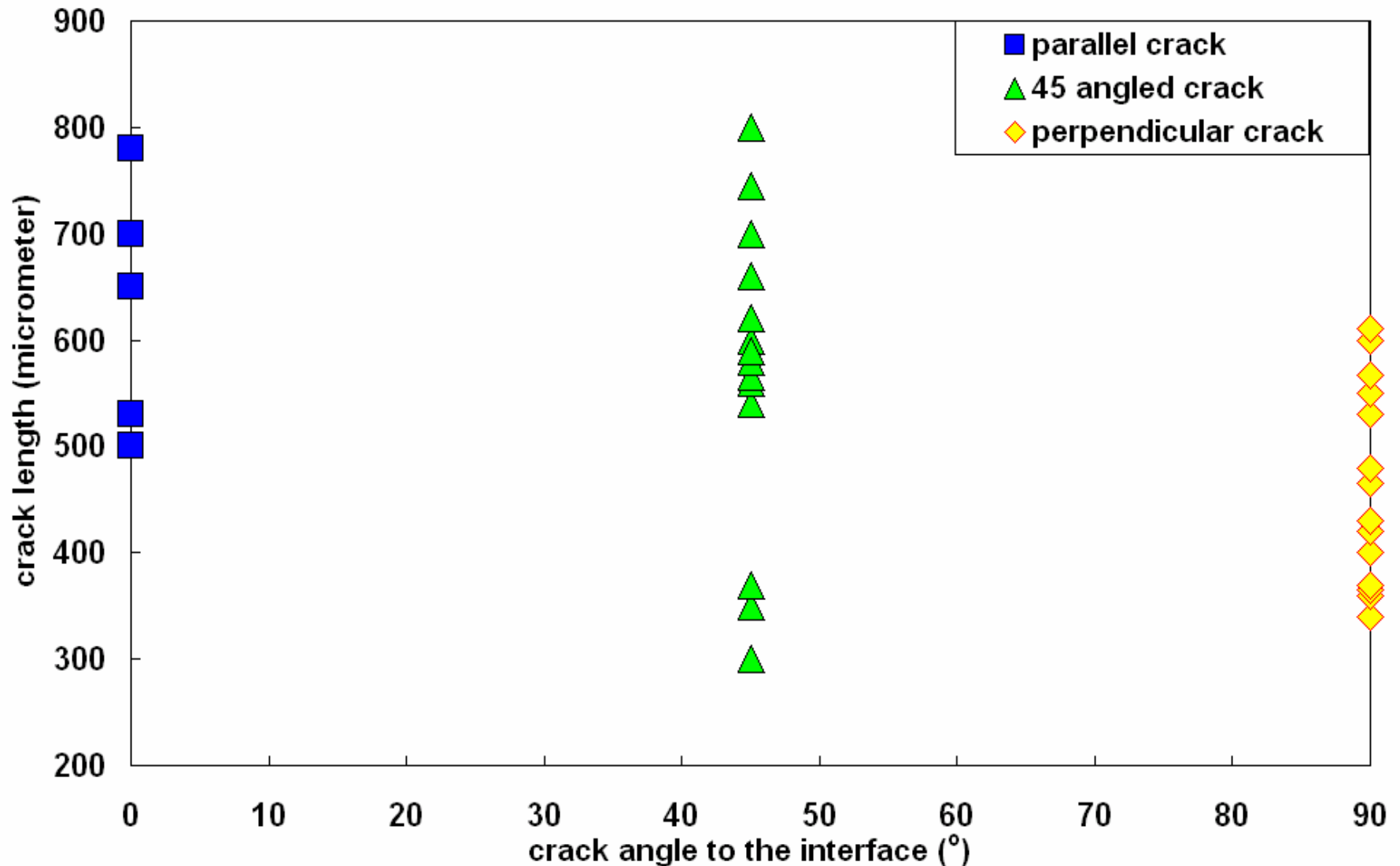
**Micro-defects along the interface**

## *Micro-defects in Untested Ti-Al<sub>3</sub>Ti*

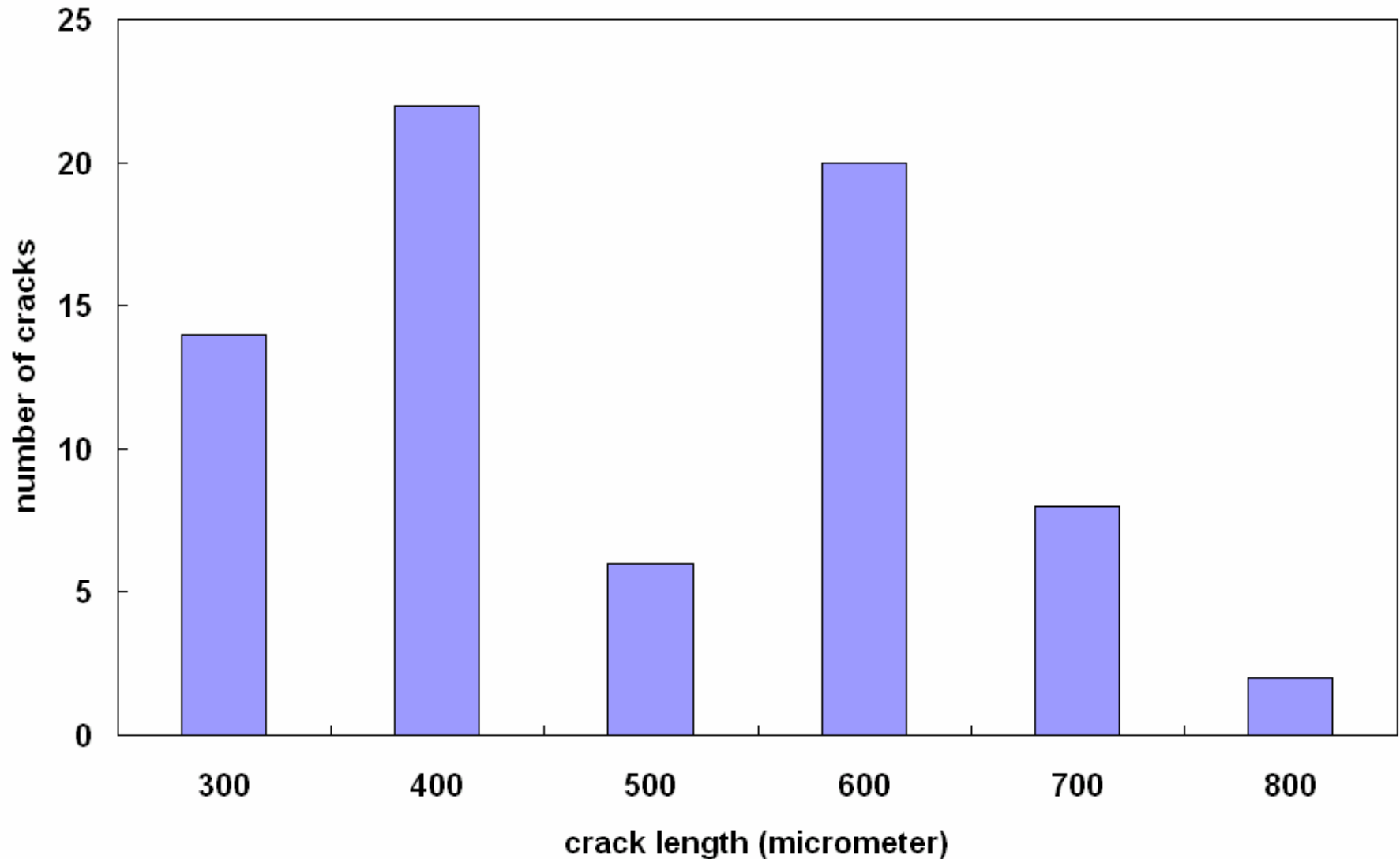


**Middle line in Al<sub>3</sub>Ti layer**

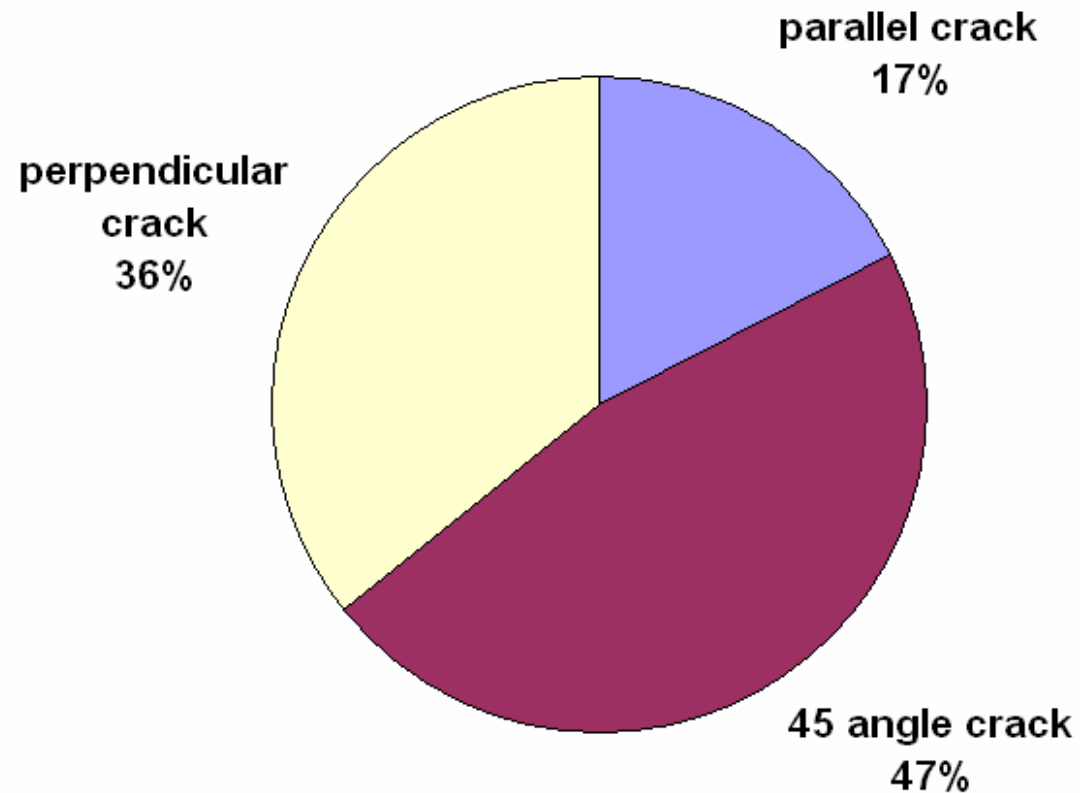
## *Crack Morphology in Untested Ti-Al<sub>3</sub>Ti I*



## *Crack Morphology in Untested Ti-Al<sub>3</sub>Ti II*



## *Crack Morphology in Untested Ti-Al<sub>3</sub>Ti III*



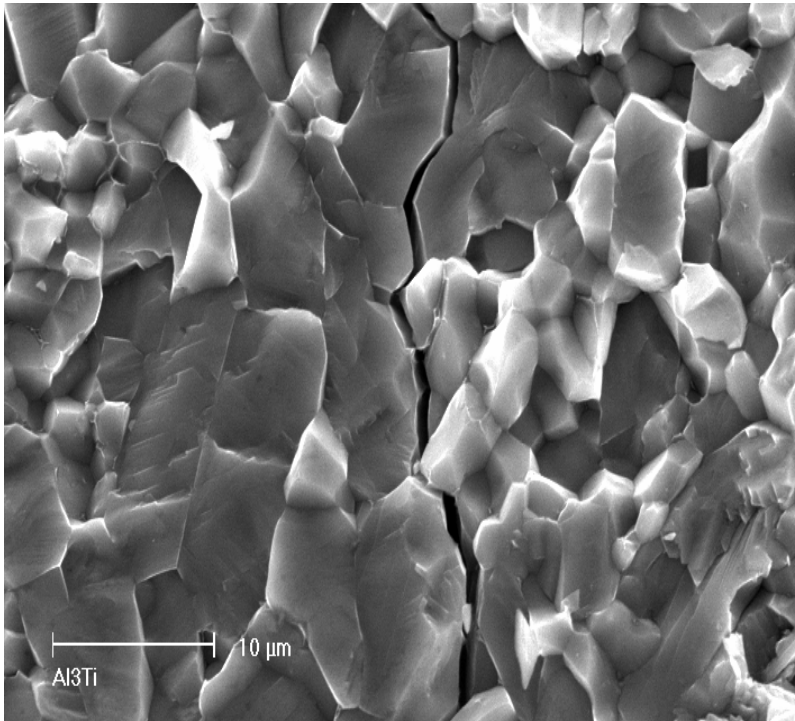
$$\rho_{crack} = \frac{\ell}{\Theta} = 1.6/cm$$

## *Compression Tests on Pure Al<sub>3</sub>Ti I*

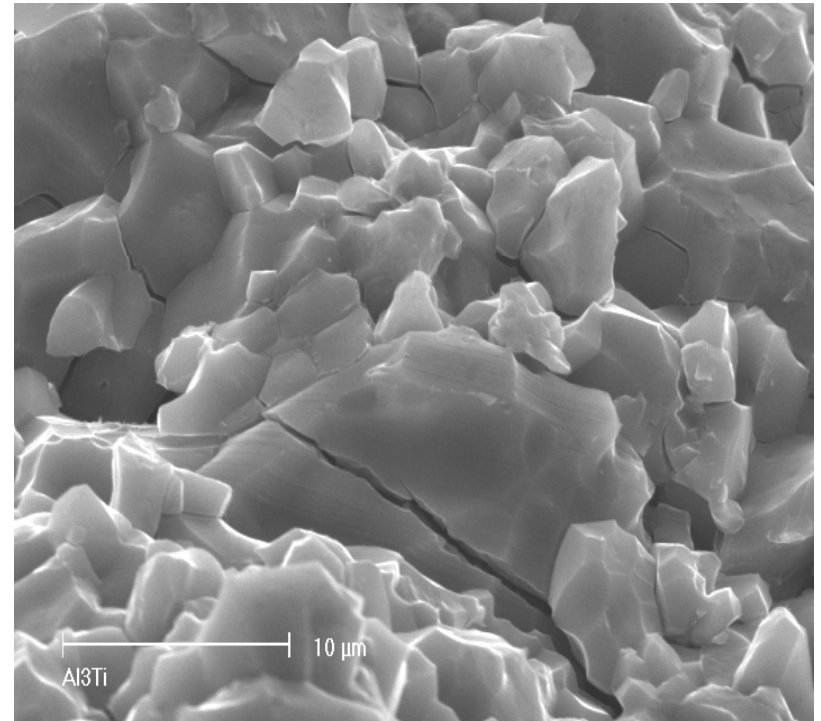
	Maximum compressive stress of pure Al <sub>3</sub> Ti (MPa)	Maximum compressive stress of Ti-Al <sub>3</sub> Ti (35% Ti, perpendicular loading) (MPa)
Dynamic (1000/s)	1285	1300
Quasistatic (0.01/s)	921	1100
Quasistatic (0.0001/s)	890	1000



## *Compression Tests on Pure $\text{Al}_3\text{Ti}$ II*



Intergranular crack



Transgranular crack

Strain rate has little influences on crack modes.

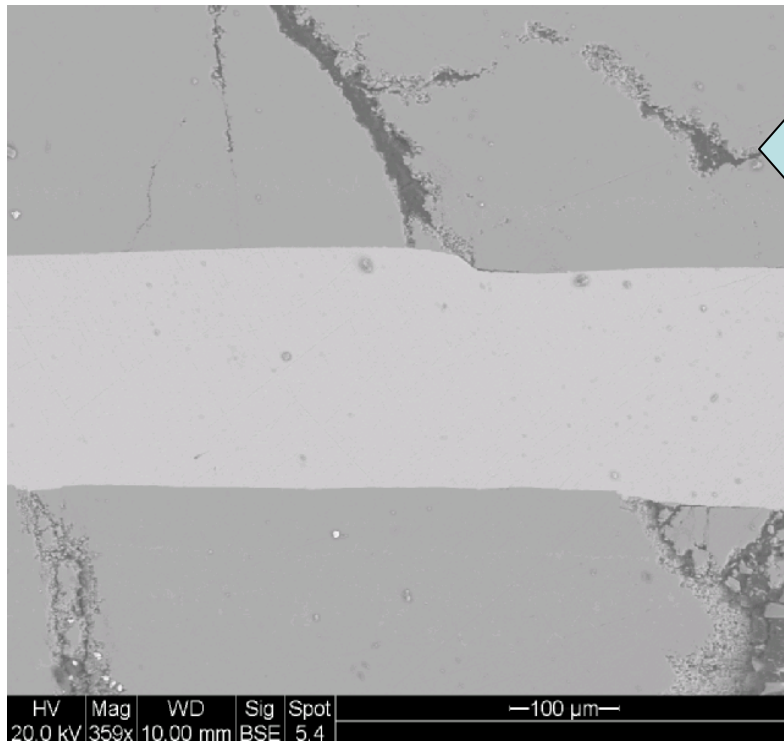
## *Compression Tests on Ti-Al<sub>3</sub>Ti I*

Volume Fraction of Ti-6-4	Loading Direction to the Laminate Plane	Strain Rate (/s)
14%	Perpendicular	0.0001
14%	Perpendicular	2800
14%	Parallel	0.0001
14%	Parallel	0.01
14%	Parallel	2100
50%	Perpendicular	0.0001
50%	Perpendicular	1300
50%	Perpendicular	2500
50%	Parallel	0.0001

# Compression Tests on $Ti-Al_3Ti$ II

effect of volume fraction of titanium

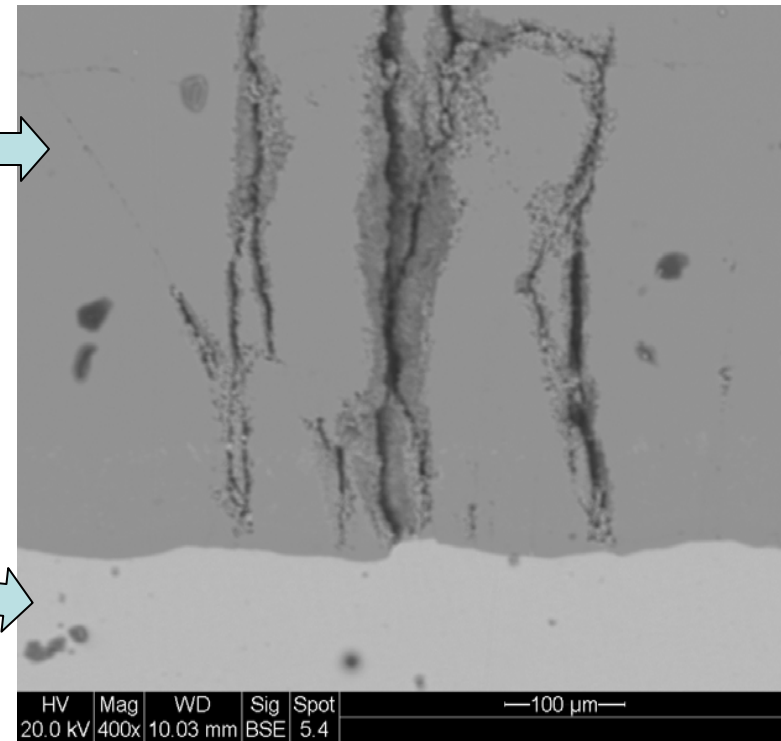
14% Ti



← Al<sub>3</sub>Ti →

50% Ti

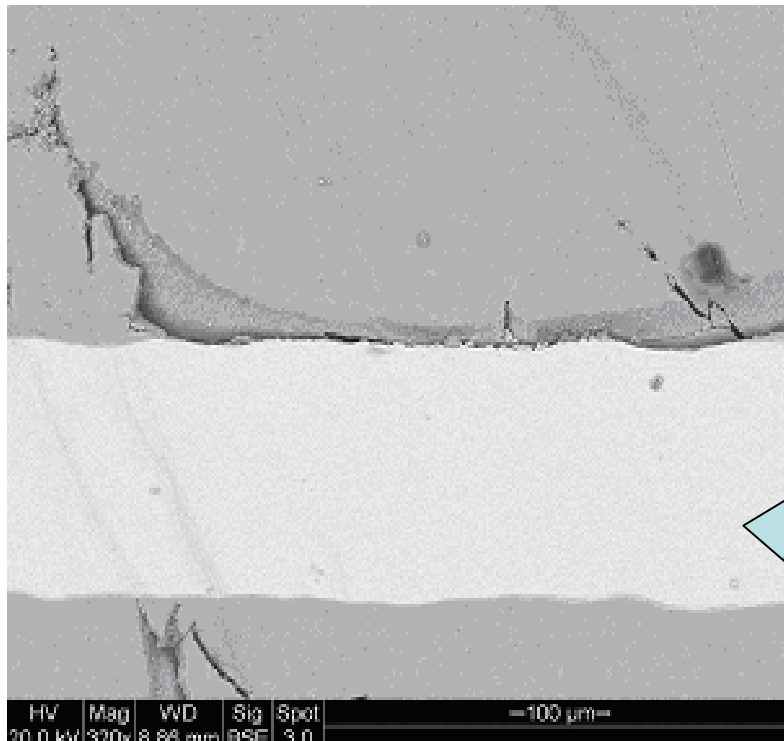
Ti →



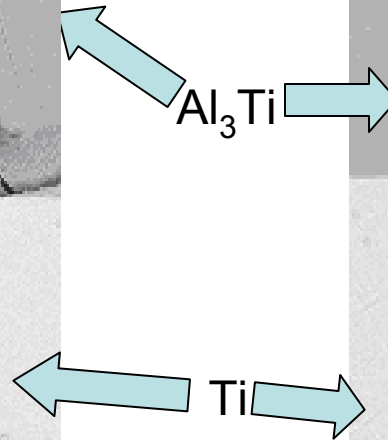
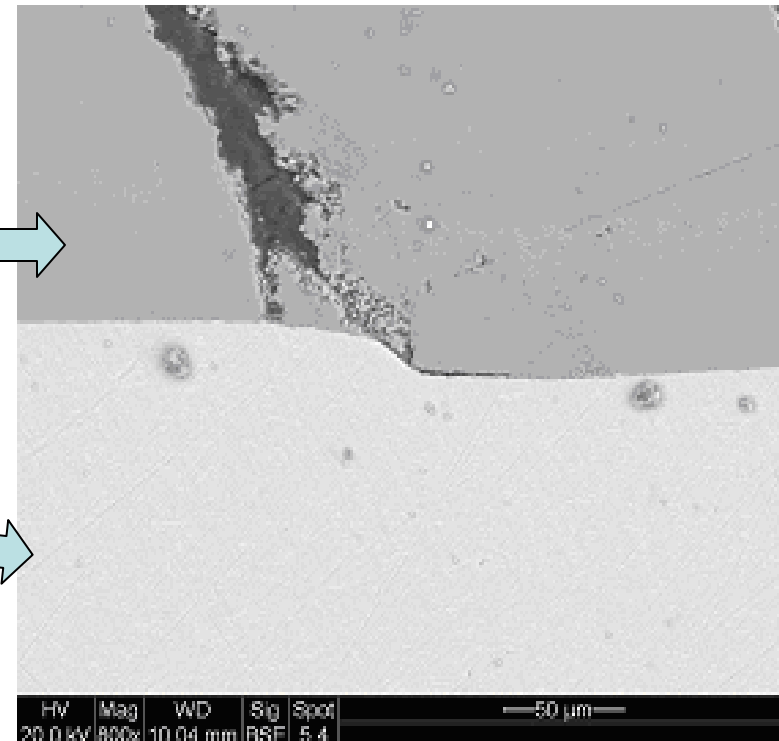
# *Compression Tests on $Ti-Al_3Ti$ III*

effect of the *strain rate*

$$\dot{\varepsilon} = 2800 / s$$

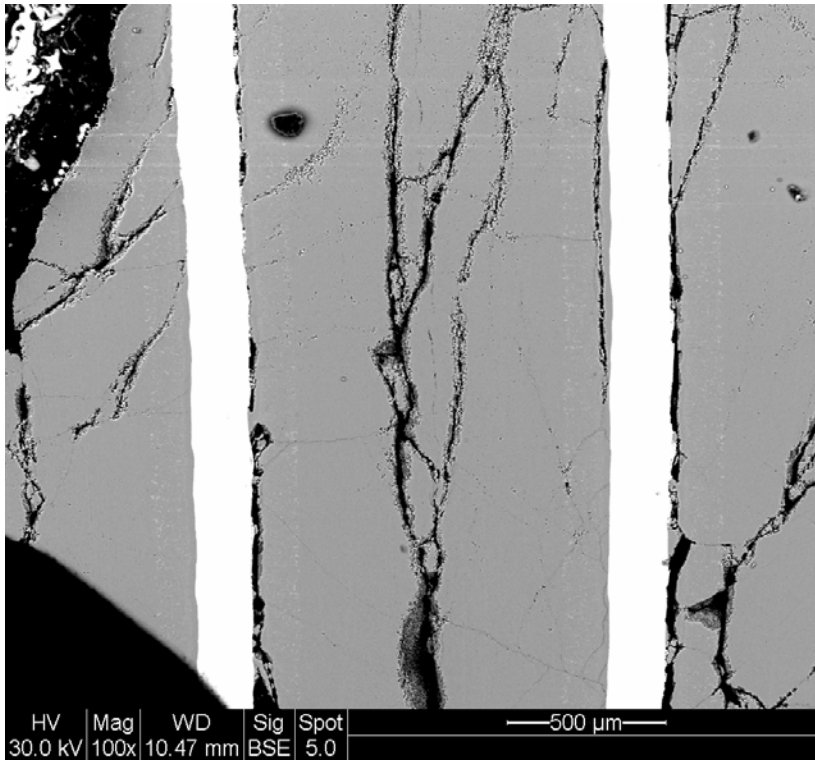


$$\dot{\varepsilon} = 0.0001 / s$$

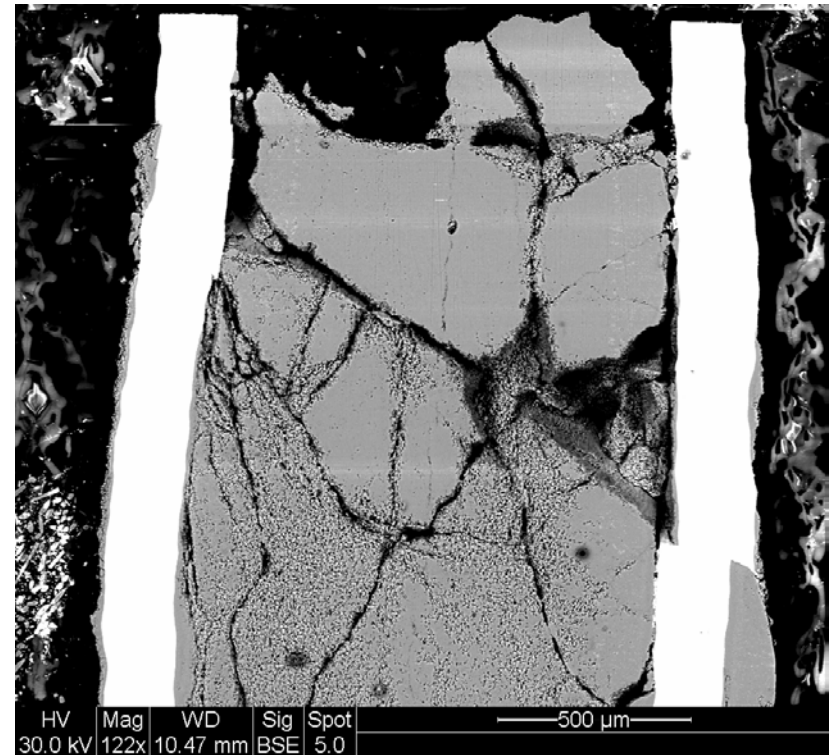


# Compression Tests on $Ti-Al_3Ti$ IV

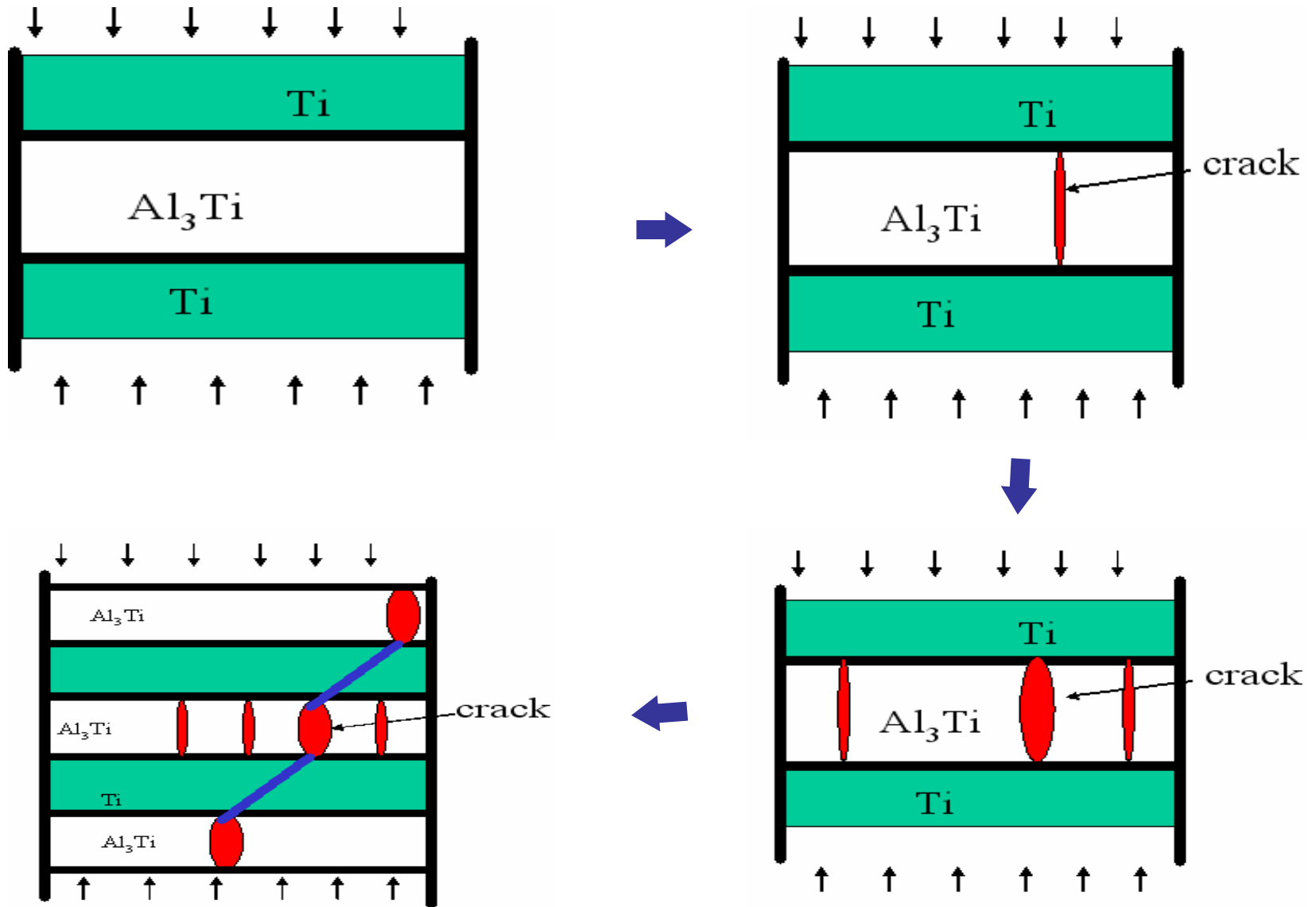
$$\dot{\varepsilon} = 2100 / s$$



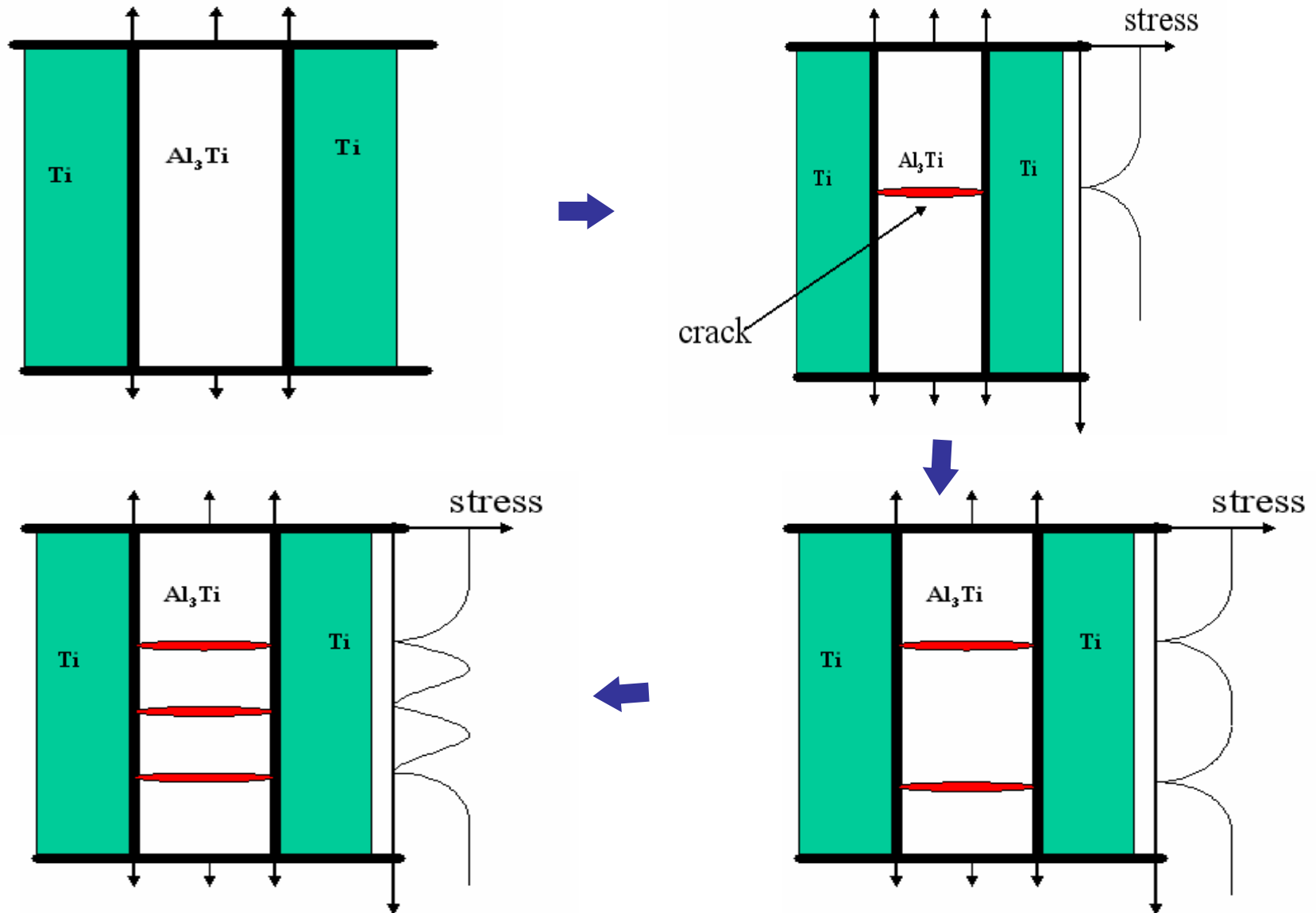
$$\dot{\varepsilon} = 0.0001 / s$$



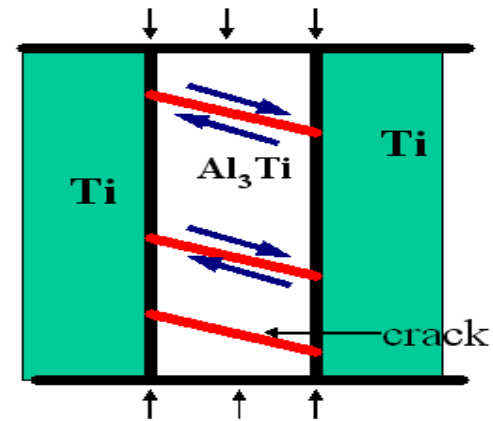
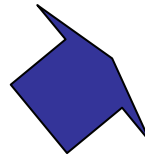
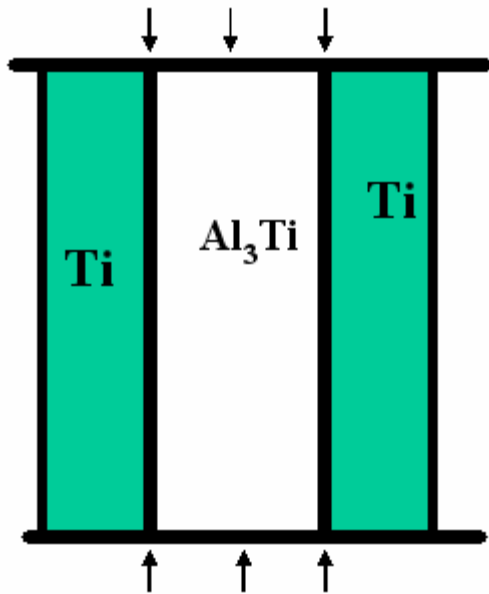
# *Physical Modeling: Damage Evolution I*



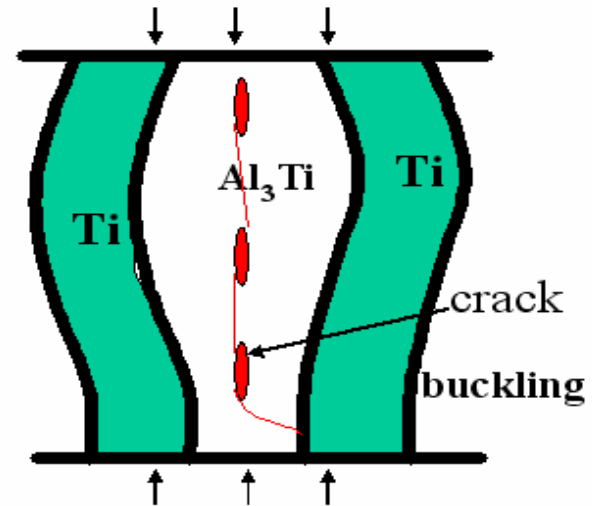
# Physical Modeling: Damage Evolution II



# *Physical Modeling: Damage Evolution III*



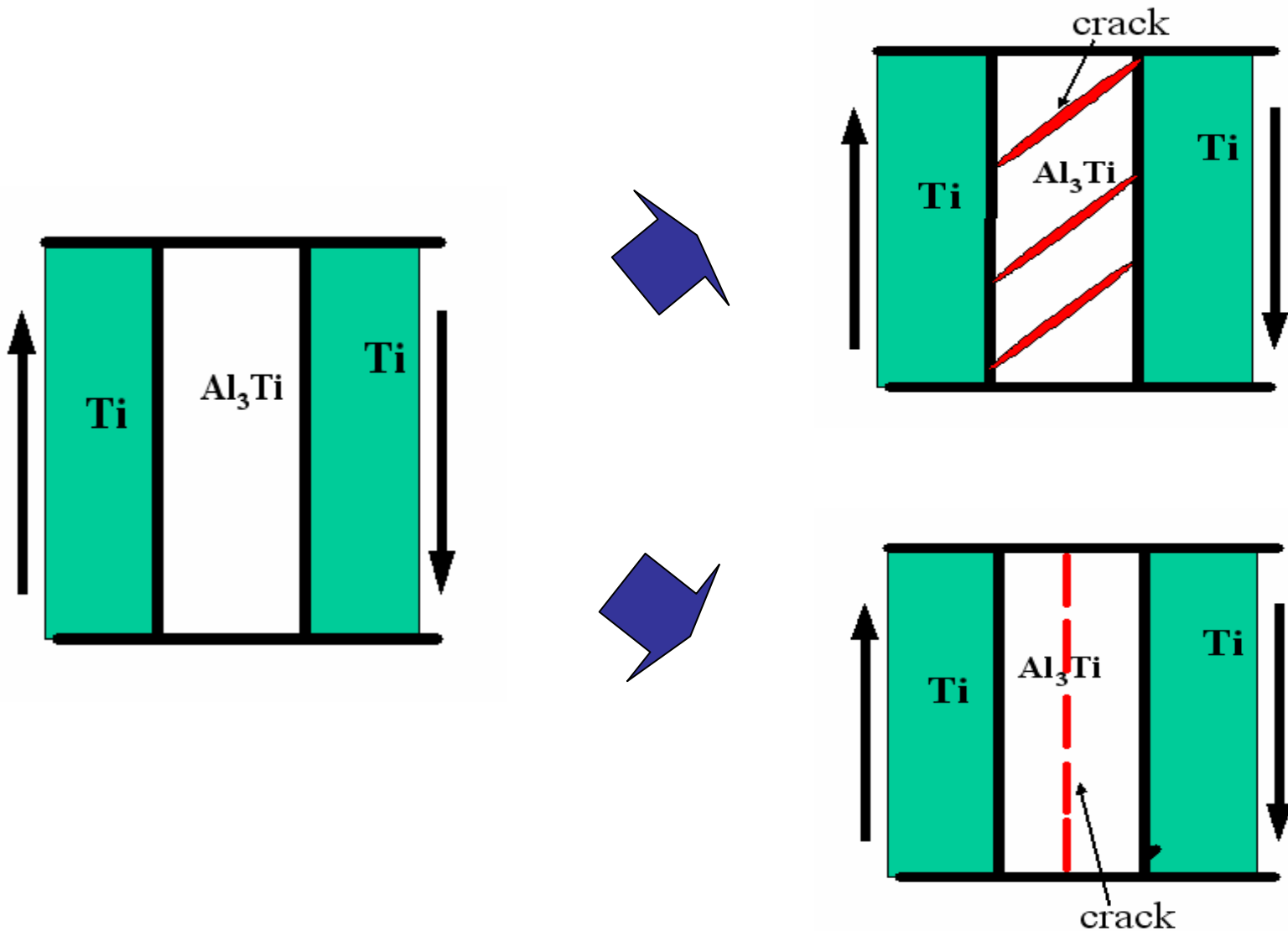
With Confinement



Without Confinement

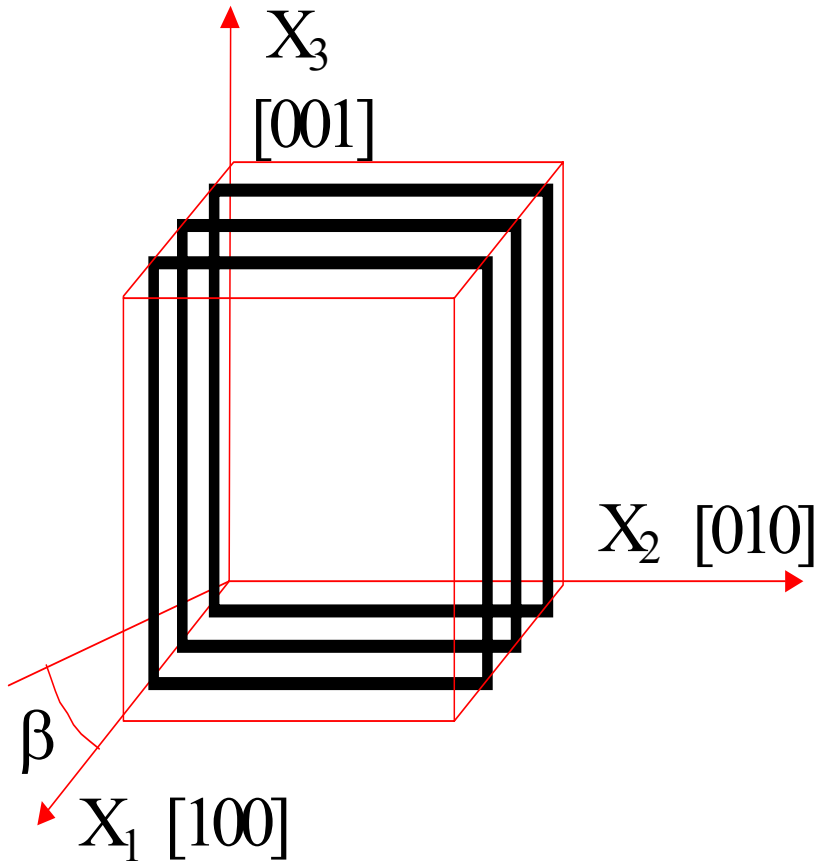


# *Physical Modeling: Damage Evolution IV*



# *Theoretical Modeling*

# *Elastic Properties of $\text{Ti-Al}_3\text{Ti}$*

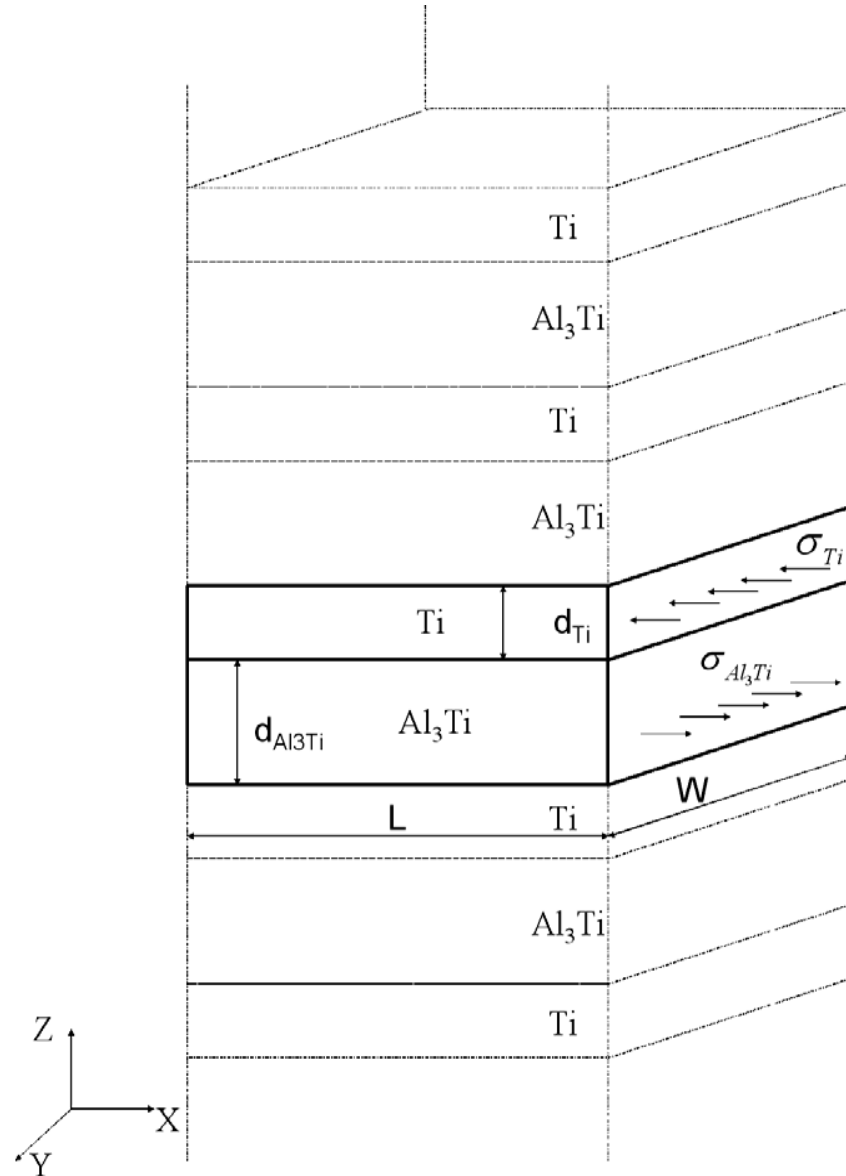


● Volume Fraction

● Orientation

# *Residual Stress in Ti-Al<sub>3</sub>Ti*

	Ti	Al <sub>3</sub> Ti
$\alpha (10^{-6}/^{\circ}\text{C})$	9.5	13



## *Residual Stress in Ti-Al<sub>3</sub>Ti*

Calculated residual stress

$$\sigma_c = \frac{E_1(1-c) \cdot [(\alpha_2 - \alpha_1) \cdot \Delta T]}{\left[ 1 + \left( \frac{1-\gamma_2}{1-\gamma_1} \cdot \frac{E_1}{E_2} - 1 \right) \cdot c \right] (1-\gamma_1)}$$

	14%Ti	20%Ti	35%Ti
Calculated residual stress (MPa)	345	327	282
Measured residual stress (MPa)	65.01	32.38	8.29
Measured residual stress (slow cooling) (MPa)	30.79	25.24	10.59

# *Residual Stress Release Mechanism I: Creep*

➤ Creep known as time-dependent deformation is characterized by Doner-Conrad equation

$$\frac{\dot{\varepsilon}_s kT}{D \mu b} = A \left( \frac{\sigma}{\mu} \right)^n$$

$\dot{\varepsilon}_s$  is the steady-state strain rate

$\mu$  the shear modulus

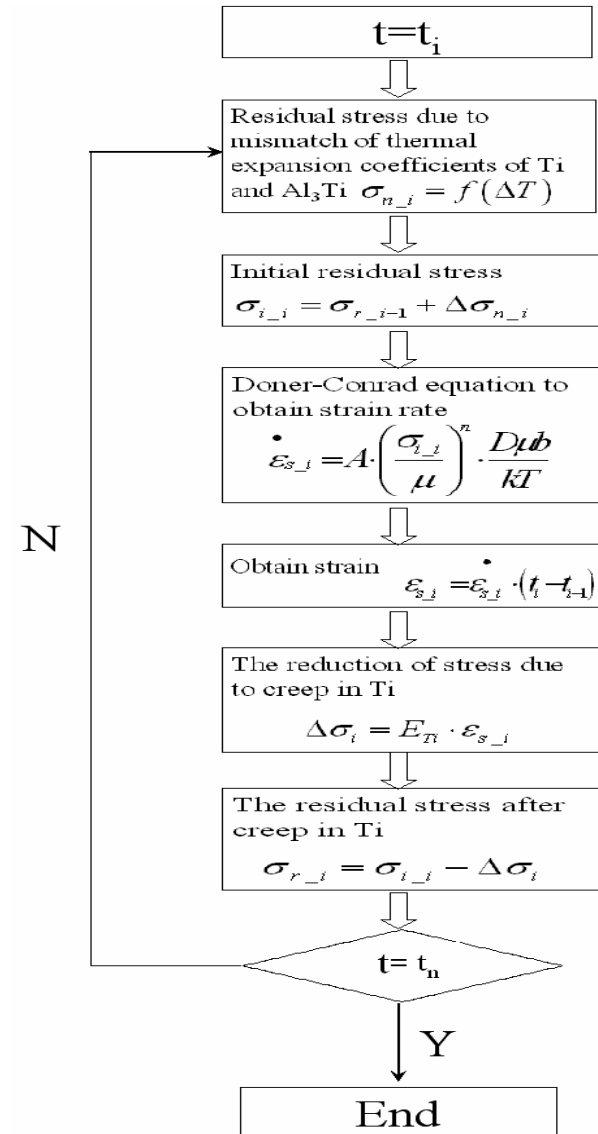
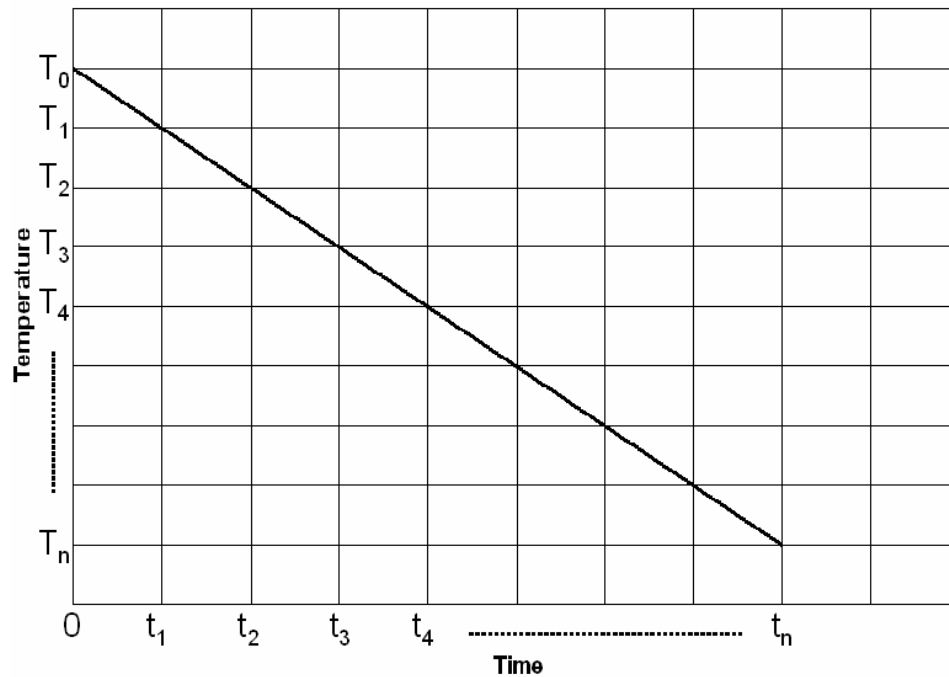
D: the diffusion coefficient

b: the Burgers vector

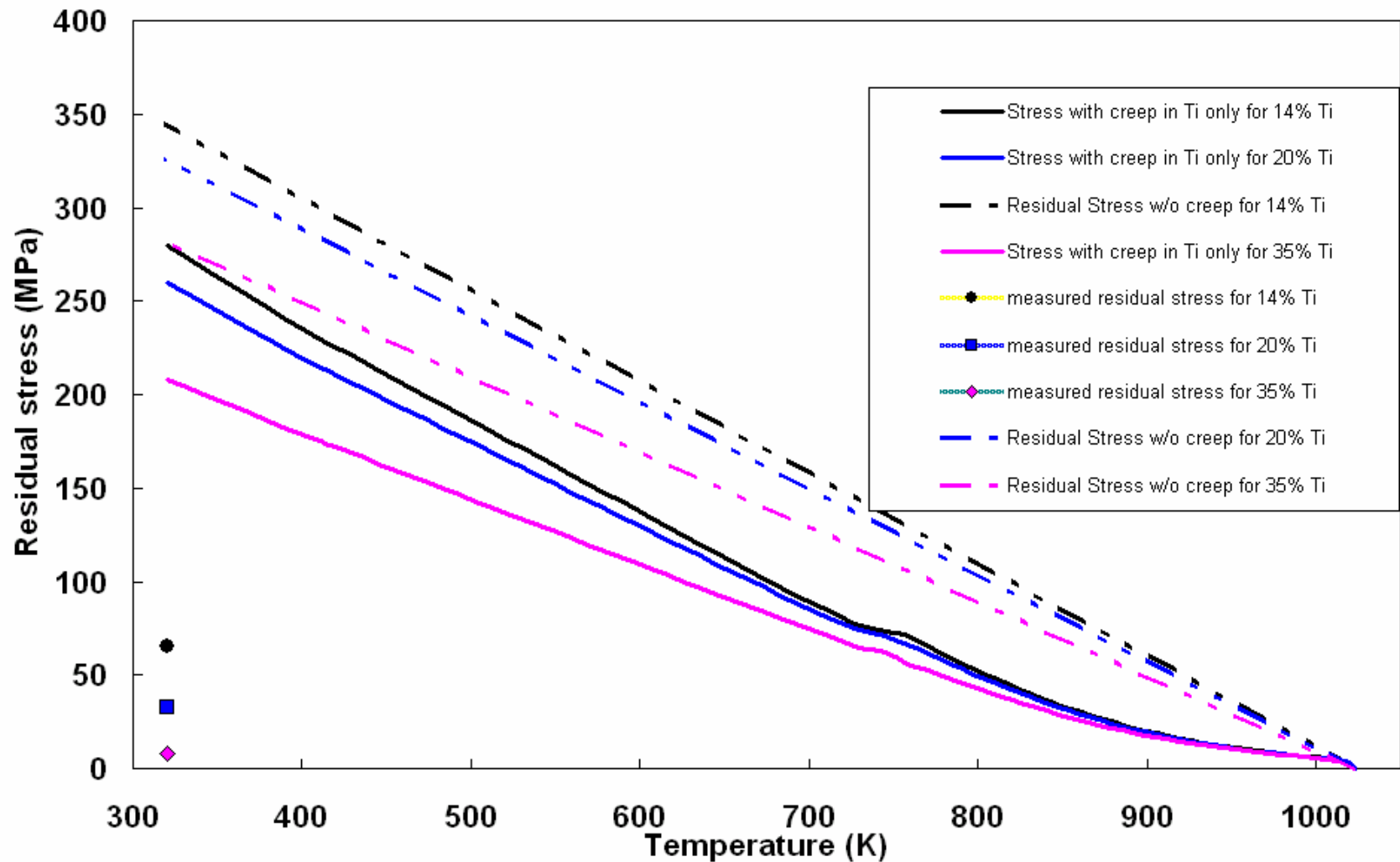
T: the temperature

k: the Boltzmann's constant

# Residual Stress Release Mechanism I: Creep

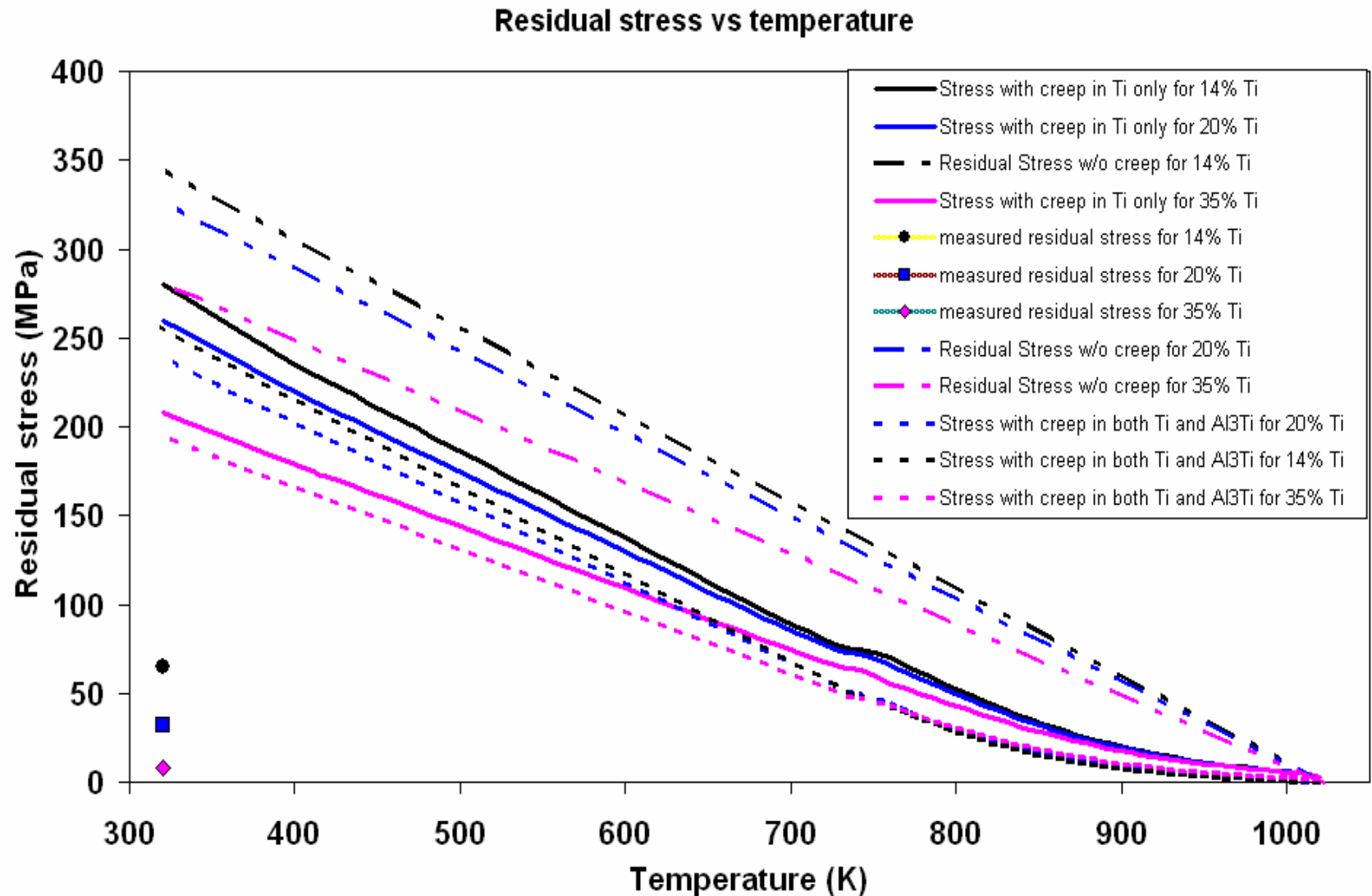


# *Residual Stress Release Mechanism I: Creep in Ti*

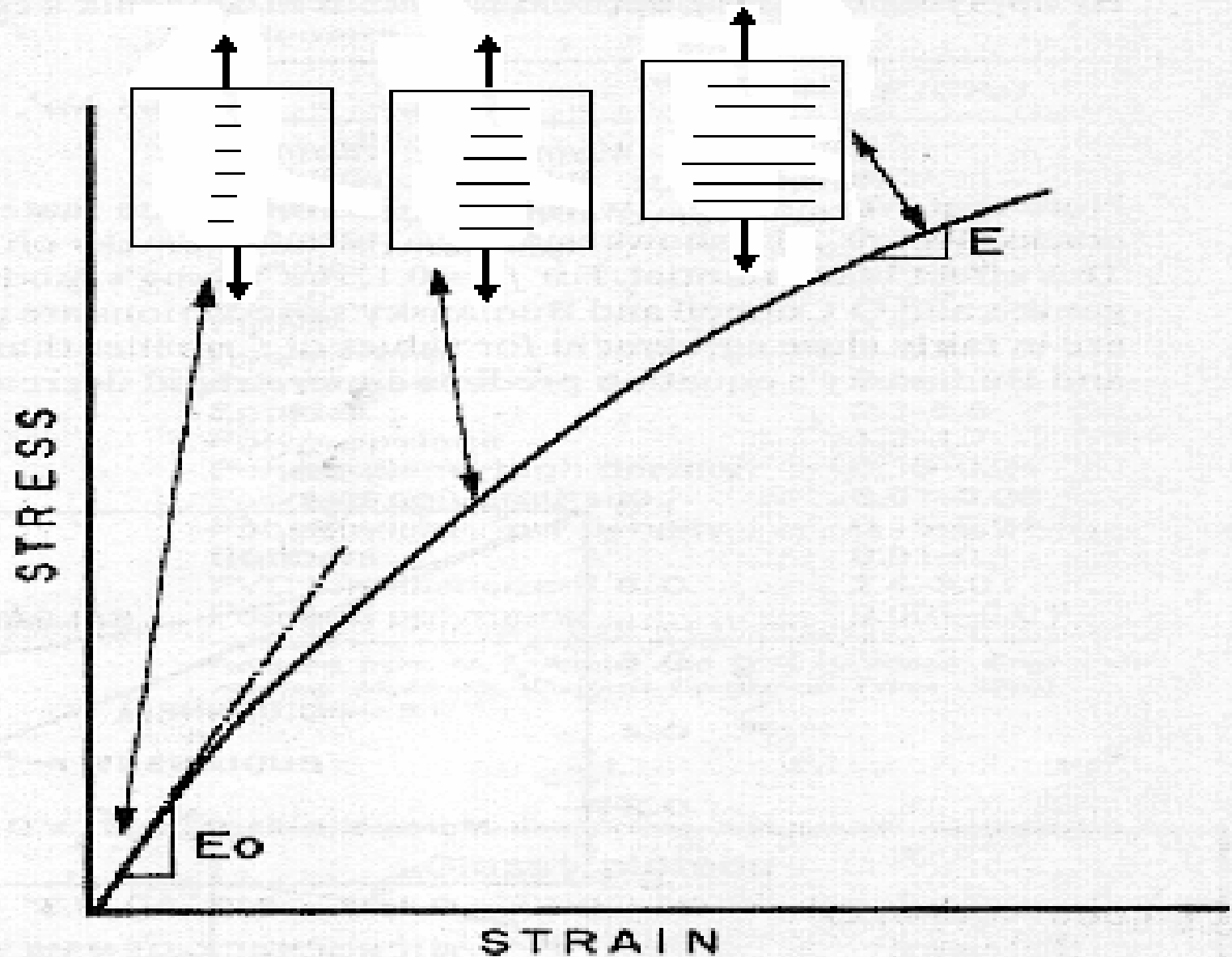




# Residual Stress Release Mechanism I: Creep in Both Ti and Al<sub>3</sub>Ti



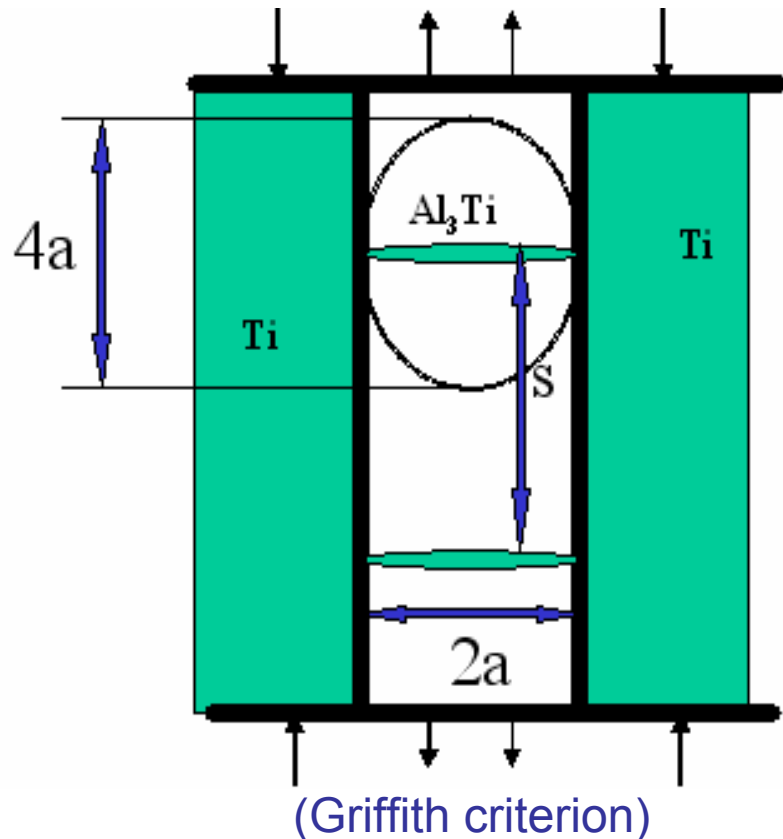
## Residual Stress Release Mechanism II: Crack Propagation



# Residual Stress Release Mechanism II: Crack Propagation

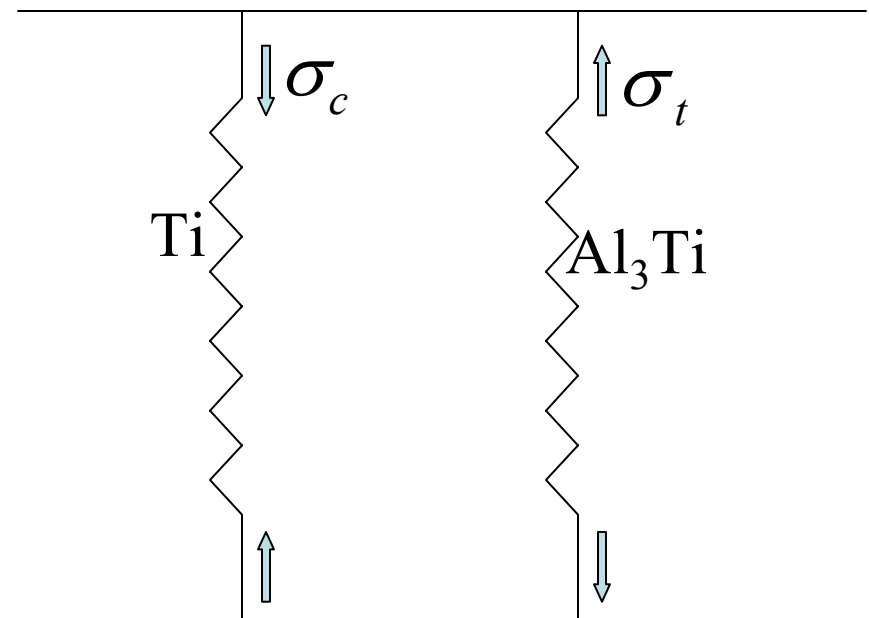
$$\sigma_c = \frac{E_1(1-c) \cdot [(\alpha_2 - \alpha_1) \cdot \Delta T]}{\left[ 1 + \left( \frac{1-\gamma_2}{1-\gamma_1} \cdot \frac{E_1}{E_2^e} - 1 \right) \cdot c \right] (1-\gamma_1)}$$

$$E_2^e = E_2 \left( a, \frac{s}{a} \right)$$



## Spring model

$$E_2^e = E_2 \cdot \left( 1 - \frac{4a}{s} \right)$$

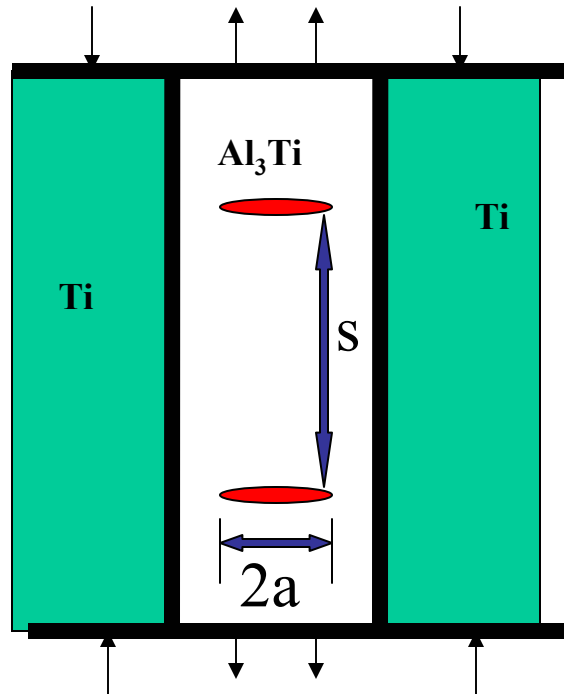


## Residual Stress Release Mechanism II: Crack Propagation

$$\sigma_c = \frac{E_1 (1 - c) \cdot [(\alpha_2 - \alpha_1) \cdot \Delta T]}{\left[ 1 + \left( \frac{1 - \gamma_2}{1 - \gamma_1} \cdot \frac{E_1}{E_2^e} - 1 \right) \cdot c \right] (1 - \gamma_1)}$$

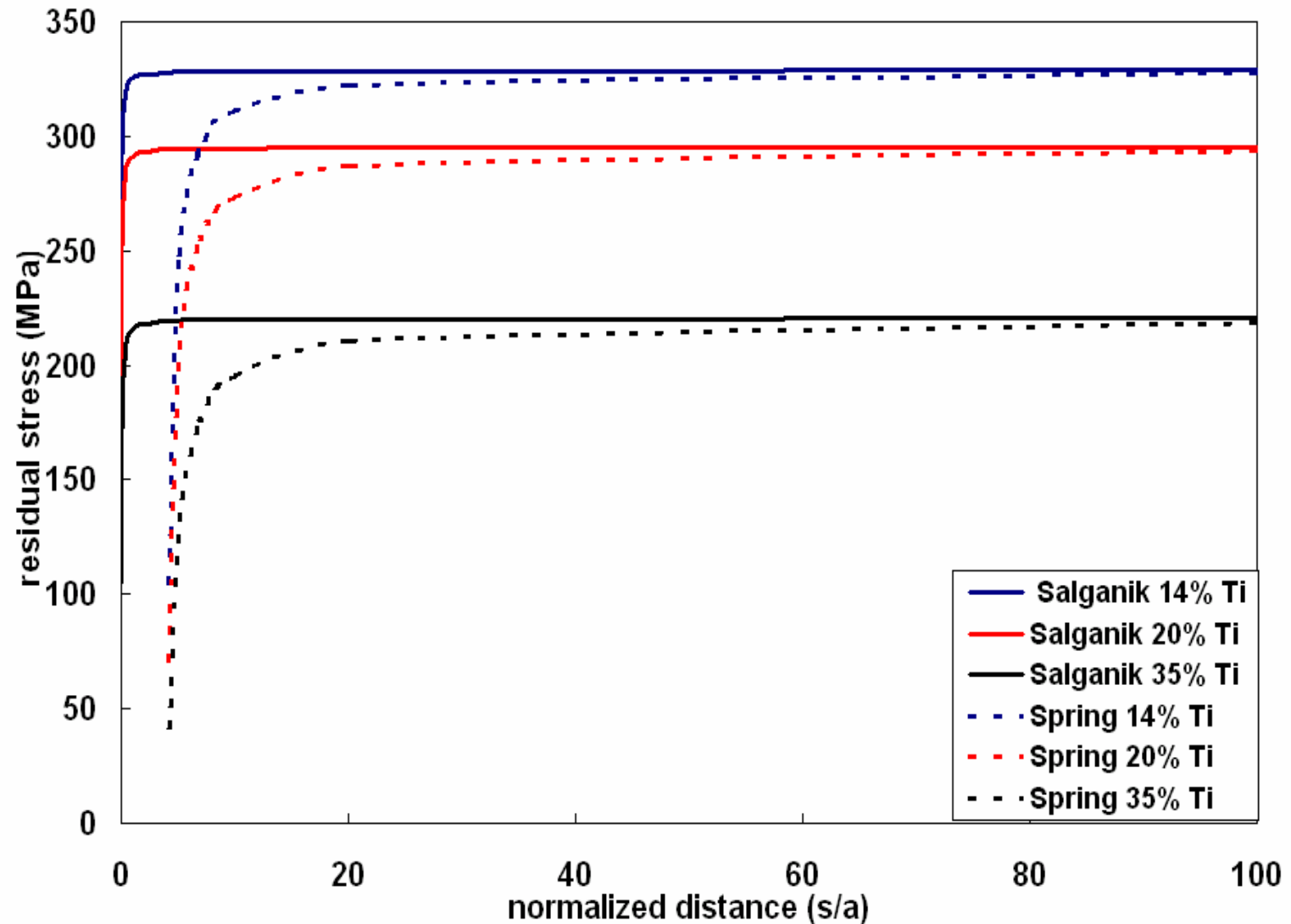
$$E_2^e = E_2 \left( a, \frac{s}{a} \right)$$

### Salganik model



$$E_2^e = \frac{E_2}{\left( 1 + \frac{16 \cdot (10 - 3\nu_2) \cdot (1 - \nu_2^2)}{45 \cdot (2 - \nu_2)} \cdot N \left( a, \frac{s}{a} \right) \cdot a^3 \right)}$$

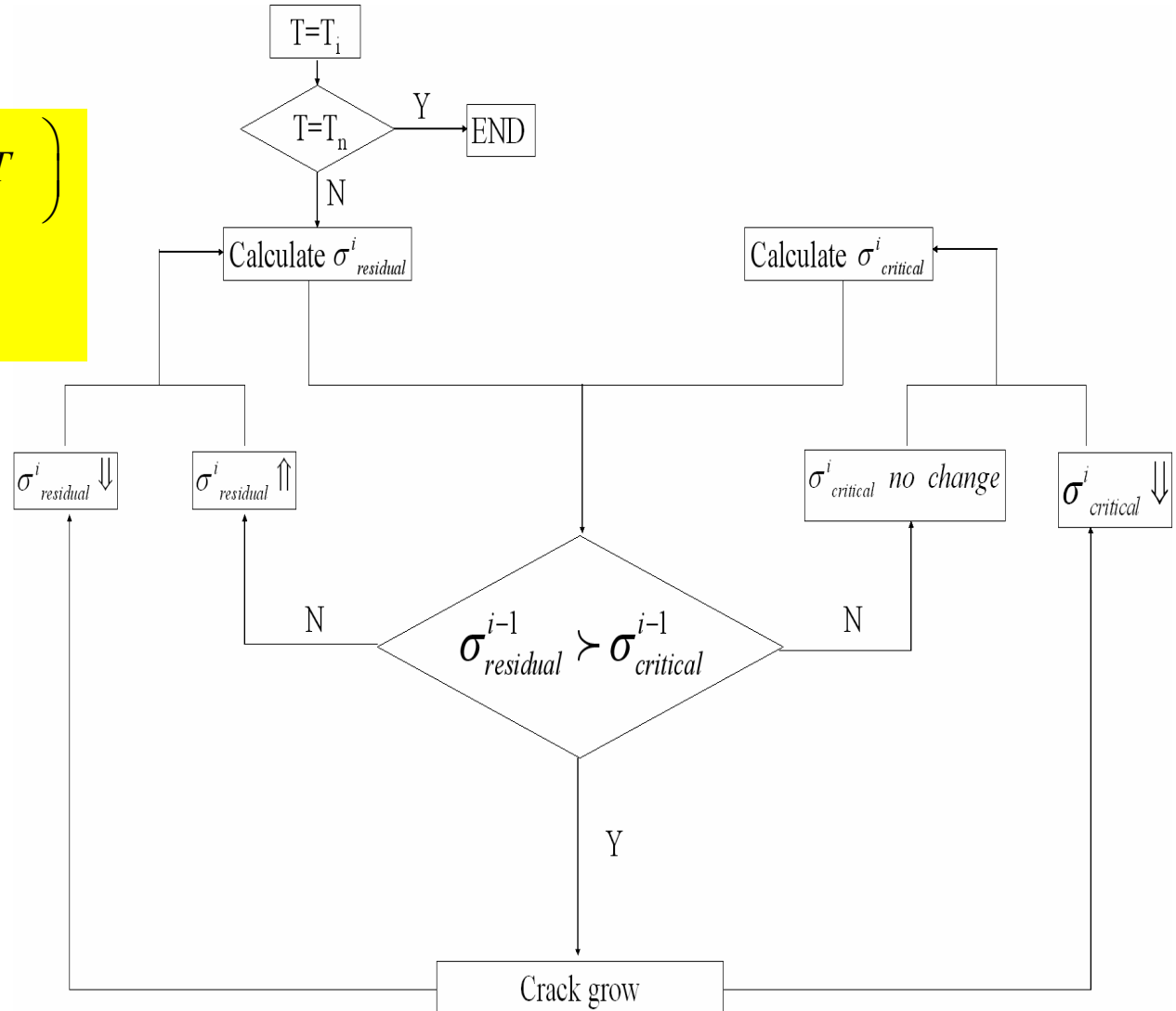
## *Residual Stress Release Mechanism II: Crack Propagation*



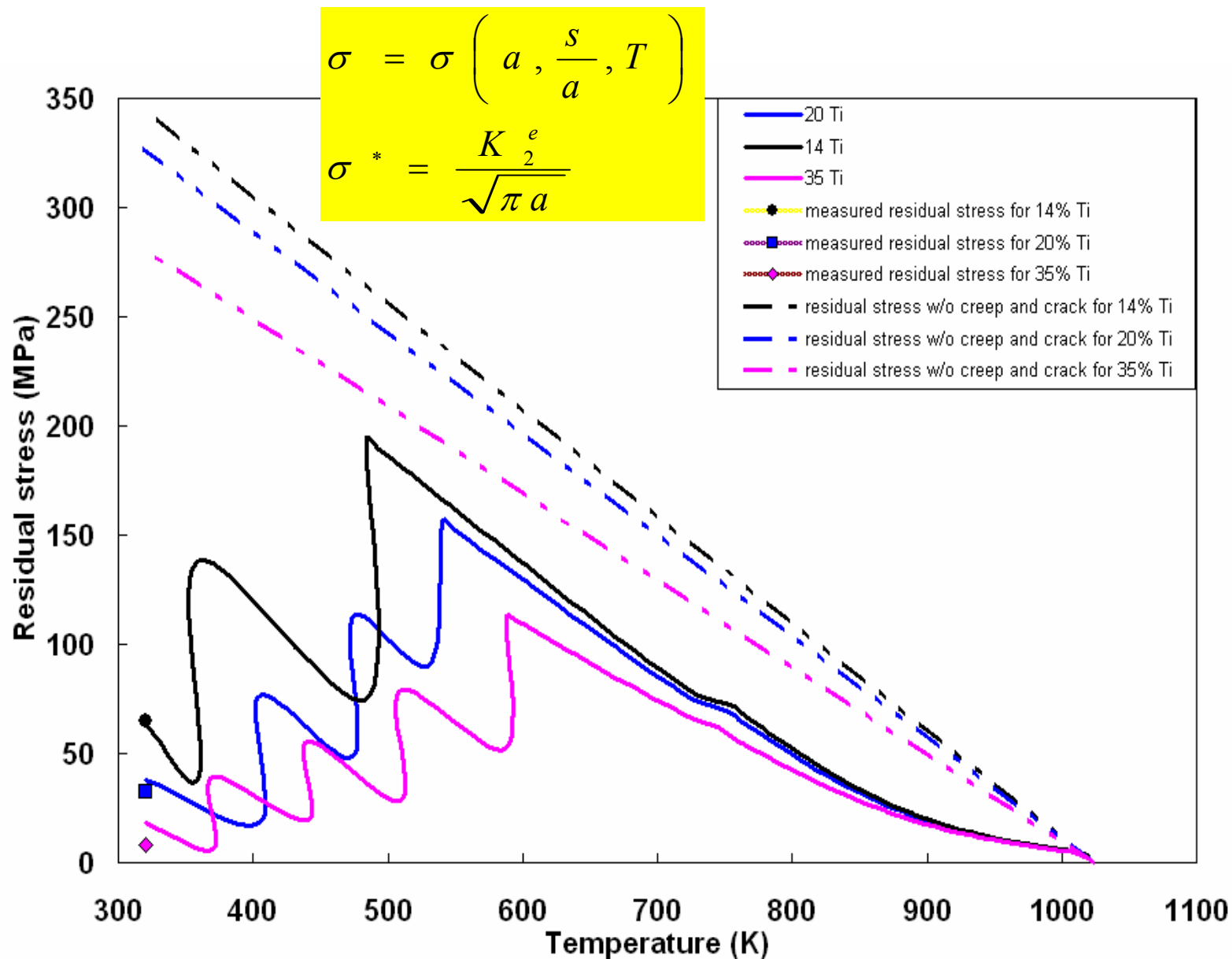
# *Residual Stress Release Mechanism: Combining Creep and Crack Propagation*

$$\sigma_{residual} = \sigma \left( a, \frac{s}{a}, T \right)$$

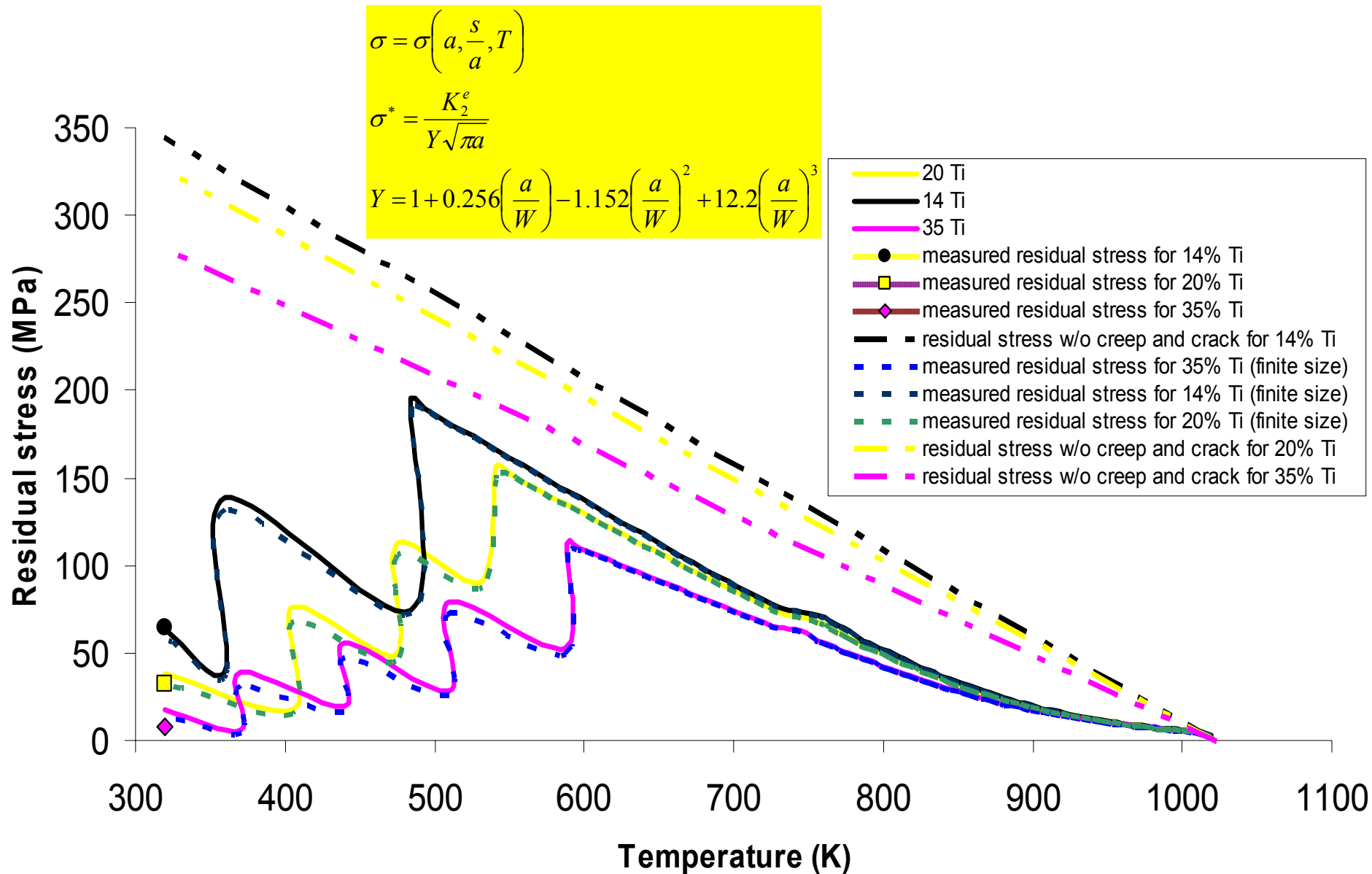
$$\sigma^*_{critical} = \frac{K_2^e}{\sqrt{\pi a}}$$



# *Residual Stress Release Mechanism*



# *Residual Stress Release Mechanism (cont'd)*





# Fracture Toughness

Apparent fracture toughness of Ti-Al<sub>3</sub>Ti MIL composite can be calculated from

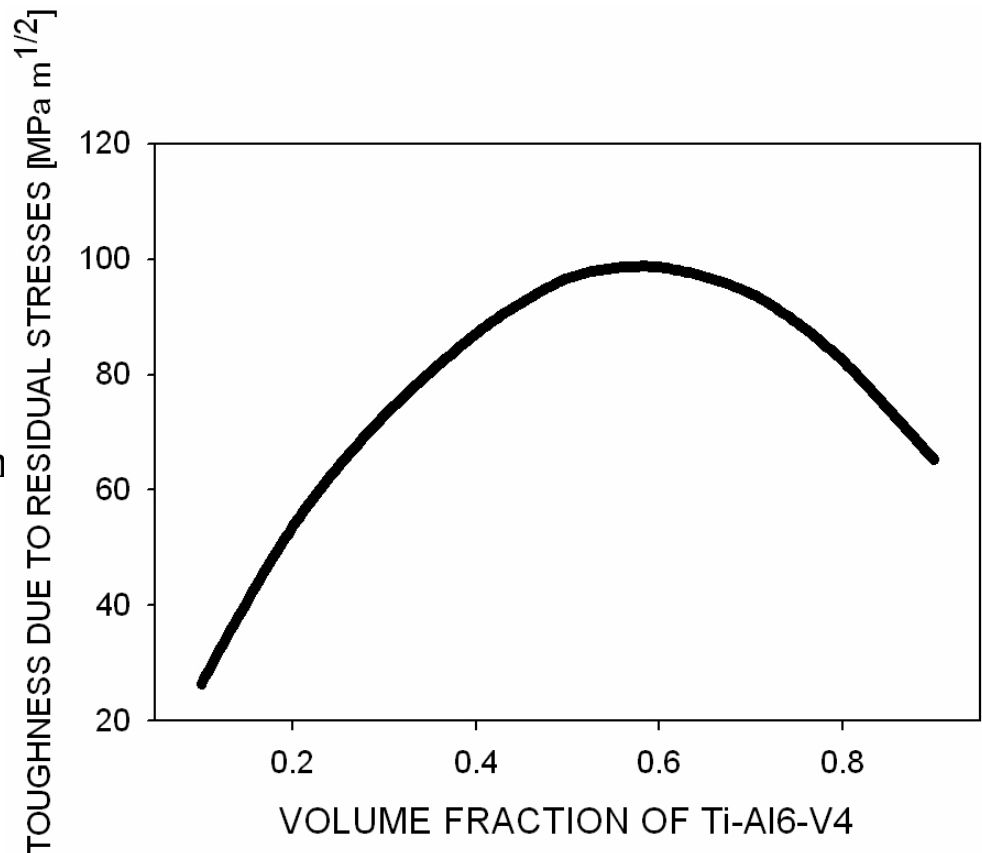
$$K_{c, app} = K_T + K_0$$

stress intensity due to the designed residual stresses

crack-initiation toughness of MIL

$$K_T = \int_0^a h\left(\frac{x}{a}, \alpha\right) \cdot \sigma_r(x) dx$$

Stress intensity was evaluated as a function of volume fraction of Ti:



## *Preliminary Conclusions*

## *Preliminary Conclusions (Experiments)*

- Optical microscopy observations have showed the crack morphology of untested Ti-Al<sub>3</sub>Ti MIL composites.
- Quasi-static and dynamic compression tests were conducted on pure Al<sub>3</sub>Ti. Transgranular and intergranular cracks were found under SEM. The experiments show the strain rate has little effects on the formation of the two different crack morphologies.
- Compression tests were conducted on the Ti-Al<sub>3</sub>Ti MIL composites, and the effects of different loading directions, different strain rates and different volume fraction of titanium on the damage evolution have been assessed.
- Four different damage evolution mechanisms have been identified, including axial splitting, shear localization in Ti layers, crack propagation along the central plane of weakness in Al<sub>3</sub>Ti, and delamination at Al<sub>3</sub>Ti-Ti interface.

## *Preliminary Conclusions (Theoretical Modeling)*

- The effect of orientation and volume fraction of Ti on the elastic properties of Ti-Al<sub>3</sub>Ti has been investigated.
- Residual stress in Ti-Al<sub>3</sub>Ti has been evaluated by elastic analysis and two stress relaxation mechanisms.
- The apparent fracture toughness of Ti-Al<sub>3</sub>Ti has been evaluated using a weight function method.

***Thank you.***

# Calculations (Lines) and Data from RUS-Measurements (Dots) Elastic Constants for MIL

